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Cu(I) and Ag(I) complexes with a new type of rigid tridentate N,P,P-ligand showing highly efficient TADF and remarkable OLED performance

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3 **Cu(I) and Ag(I) complexes with a new type of rigid tri-dentate N,P,P-ligand**
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6 **showing highly efficient TADF and remarkable OLED performance**
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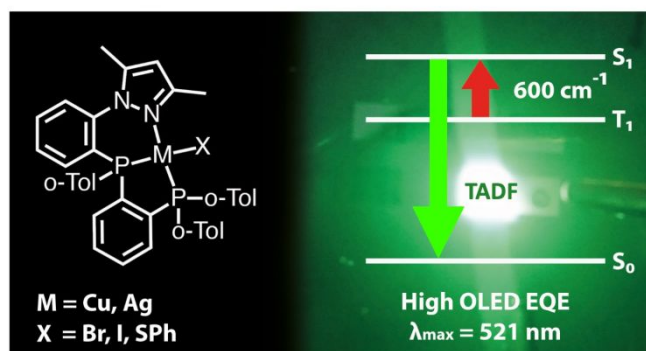
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TOC



TOC text

Neutral blue to yellow brightly luminescent copper and silver complexes with a new rigid tri-dentate N,P,P ligand (dmpzpp) were prepared and their crystal structures, TD-DFT electronic structures, as well as their phosphorescence and thermally activated delayed fluorescence (TADF) properties studied in detail between $1.7 \leq T \leq 300 \text{ K}$, giving $\Delta E(S_1-T_1)$, $k(S_1-S_0)$, $\tau(\text{phos})$, $k^{\text{nr}}(\text{phos})$, $\tau(\text{TADF})$, $k^{\text{nr}}(\text{TADF})$, $\Delta E(\text{ZFS})$, and triplet substate decay times. Sublimable $\text{Cu}(\text{dmpzpp})\text{I}$ was applied for OLED fabrication affording EQE of 16.4%.

Abstract

Neutral $\text{Cu}(\text{I})$ and $\text{Ag}(\text{I})$ complexes with a new rigid tri-dentate N,P,P ligand (dmpzpp, 3,5-dimethyl-1-(2-((2-(di-*o*-tolyl)phosphanyl)(*o*-tolyl)-phosphanyl)phenyl)-1*H*-pyrazole), giving $\text{Cu}(\text{dmpzpp})\text{Cl}$ **6**, $\text{Cu}(\text{dmpzpp})\text{Br}$ **7**, $\text{Cu}(\text{dmpzpp})\text{I}$ **8**, $\text{Cu}(\text{dmpzpp})\text{SPh}$ **9**, and $\text{Ag}(\text{dmpzpp})\text{I}$ **10** with SPh = thiophenolato, were prepared and their crystal structures, TD-DFT electronic structures, and phosphorescence as well as thermally activated delayed fluorescence (TADF) properties studied in detail. The photo-luminescence quantum yields Φ_{PL} lie between 70 and 90% with emission colors from blue to yellow. **9** with very bulky ligands showing $\Phi_{\text{PL}} = 90\%$ was used for detailed emission studies from $T = 1.7$ to 300 K. Up to $T \approx 70 \text{ K}$, **9** shows

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3 only long-lived phosphorescence with a radiative decay time of T_1 of $\tau^r(\text{phos}) = 1$ ms due to
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5 weak spin-orbit coupling. Accordingly, the zero-field splittings of T_1 in three substates is $<$
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7 1cm^{-1} (0.1 meV) with individual decay times of 2400, 2250, and 292 μs . Presumably, the
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9 phosphorescence is essentially induced by spin-vibronic mechanisms. Up to $T = 300$ K, the
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11 radiative decay time decreases by more than two orders of magnitude to $\tau^r(\text{TADF}) = 5.6$ μs due
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13 to the TADF effect. This short decay time is determined by the small gap of $\Delta E(S_1-T_1) = 600$
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15 cm^{-1} (74 meV) and the fast radiative $S_1 \rightarrow S_0$ rate of $1.1 \cdot 10^7$ s^{-1} (91 ns). For fabrication of an
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17 OLED device, we applied sublimable **8** using a co-host device structure and a concentration of
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19 **8** of 2 wt% resulting in a green-emitting OLED showing CIE coordinates of (0.33; 0.52), high
20
21 external quantum efficiency of up to EQE = 16.4%, and high luminance of almost 10000 cd
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23 m^{-2} . Strategies for designing compounds giving higher EQE are presented.

31 Introduction

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34 Fundamental research of organo-transition metal compounds was strongly stimulated by their
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36 potential or meanwhile already realized commercial application as emitters in organic light
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38 emitting diodes (OLEDs). As a consequence, extensive material research was triggered and
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40 important new compounds and exciton harvesting mechanisms were developed. Hence, a large
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42 number of materials was characterized photophysically as well as with respect to their OLED
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44 use.¹⁻¹⁶ For an efficient OLED device, it is essential that 100% of the singlet (25%) and triplet
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46 (75%) excitons^{5,9,17,18} formed in the emission layer are harvested and converted into light. This
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48 is only successful with adequately designed emitter compounds. The original OLED concept¹⁹
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50 that, based on “normal” fluorescent molecules, only exploits the singlet excitons. However,
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52 different exciton harvesting mechanisms have been developed meanwhile that improve the
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54 OLED efficiency dramatically: (i) By applying Ir(III) or Pt(II) complexes, for example,
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56 showing relatively fast and efficient phosphorescence, all singlet and triplet excitons are
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3 harvested in the compounds' lowest triplet states.^{2-5,8,9,11,12,14,20-28} This mechanism is denoted as
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5 *triplet harvesting mechanism*.^{2-5,8,9,11,18,27,28} (ii) An alternative exciton harvesting mechanism²⁹
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7 is based, for example, on Cu(I), Ag(I), W(VI), Au(III) and Au(I) complexes^{1,10,16,27,30-78} or
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9 specifically designed purely organic molecules⁷⁹⁻⁸⁹ that show the molecular effect of thermally
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11 activated delayed fluorescence (TADF).⁹⁰ In an OLED emission layer, these materials can
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13 harvest all excitons in the lowest singlet and triplet states. For most compounds, the
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15 phosphorescence from the triplet state T_1 to the electronic ground state S_0 is largely forbidden.
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17 However, if the energy separation between the T_1 state and the next higher lying singlet state
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19 S_1 is small enough, for example, displaying $\Delta E(S_1-T_1) < 1000 \text{ cm}^{-1}$ (125 meV)⁴⁹ thermal
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21 activation from T_1 to the S_1 state can be efficient at ambient temperature. Hence, all excitons
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23 can be converted into light via emission from the singlet state. Therefore, this mechanism has
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25 been denoted as *TADF-singlet harvesting mechanism*. (iii) Recently, another mechanism has
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27 been proposed⁹¹⁻⁹³ by which particularly short emission decay times could be achieved. This
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29 was realized by application of specifically developed organic molecules that exhibit an almost
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31 ignorable energy separation $\Delta E(S_1-T_1)$ compared to the energy available at ambient temperature
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33 of 210 cm^{-1} (26 meV). Moreover, appropriate electronic structure design was required to
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35 guarantee fast intersystem crossing (ISC) between the S_1 and T_1 states. Accordingly, also by
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37 this mechanism all singlet and triplet excitons can be converted into light. Due to the very small
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39 energy separation and the *direct* and fast equilibration between singlet and triplet states without
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41 time-delaying TADF processes, this mechanism has been denoted as *direct singlet harvesting*
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43 (*DSH*) *mechanism*.⁹¹⁻⁹³ (iv) Very recently, the *doublet harvesting mechanism* was proposed.
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45 Applying this mechanism, also 100% of the excitons can be transferred into light, particularly,
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47 for deep blue light generation in combination with hyper-fluorescence for very fast decaying
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49 electro-luminescence.⁹⁴
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3 Each of the described exciton harvesting mechanisms requires design of specific emitter
4 materials. Moreover, they have to be optimized, for example, with respect to the emission color,
5 OLED efficiency and performance, emission band width, and especially, short decay
6 time^{49,59,62,91,95} to reduce roll-off and device stability problems.⁹⁵ Furthermore, environmental
7 compatibility and availability at reasonable costs cannot be disregarded.
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15 In this contribution, we want to focus on environmentally friendly TADF-singlet harvesting
16 Cu(I) and Ag(I) complexes. Although many investigations with such complexes have already
17 been carried out, it is still a challenge to design new materials that exhibit high emission
18 quantum yields at short TADF decay times. This is due to the typical property of Cu(I) or Ag(I)
19 TADF emitters of which the lowest excited states are of metal-to-ligand charge transfer
20 (MLCT) character. Upon excitation of these states, the metal center is formally oxidized. As a
21 consequence, the pseudo-tetrahedral ground state geometry is strongly distorted (flattened)
22 towards a planar geometry.⁹⁶⁻⁹⁸ In this situation, the Franck-Condon (FC) factors between the
23 low-lying vibrational wavefunctions of the excited states and those of the high-lying vibrations
24 of the electronic ground state become large. This is usually connected with fast non-radiative
25 decay rates and thus, small emission quantum yields.⁹⁹ Therefore, it is highly desirable to
26 synthesize rigid complexes with little freedom for distortions in the excited states. For example,
27 use of rigid chelate ligands combined with sterical hindrances between them seems to be an
28 adequate strategy. This approach was already successful for a few Cu(I)^{54,58} and Ag(I)^{62,100}
29 complexes. Here, we will investigate Cu(I) and Ag(I) complexes with a new rigid tri-dentate
30 N,P,P-ligand, namely 3,5-dimethyl-1-(2-((2-(di-*o*-tolyl)phosphanyl)(*o*-tolyl)-
31 phosphanyl)phenyl)-1*H*-pyrazole (dmpzpp). (Scheme 1, below) Indeed, the resulting
32 complexes Cu(dmpzpp)Br **7**, Cu(dmpzpp)I **8**, Cu(dmpzpp)SPh **9** (with SPh = thiophenolate),
33 and Ag(dmpzpp)I **10** show high emission quantum yields Φ_{PL} of 70% up to 90% for powder
34 materials. Even in films used as hosts for OLED fabrication, the Φ_{PL} decrease is not as drastic
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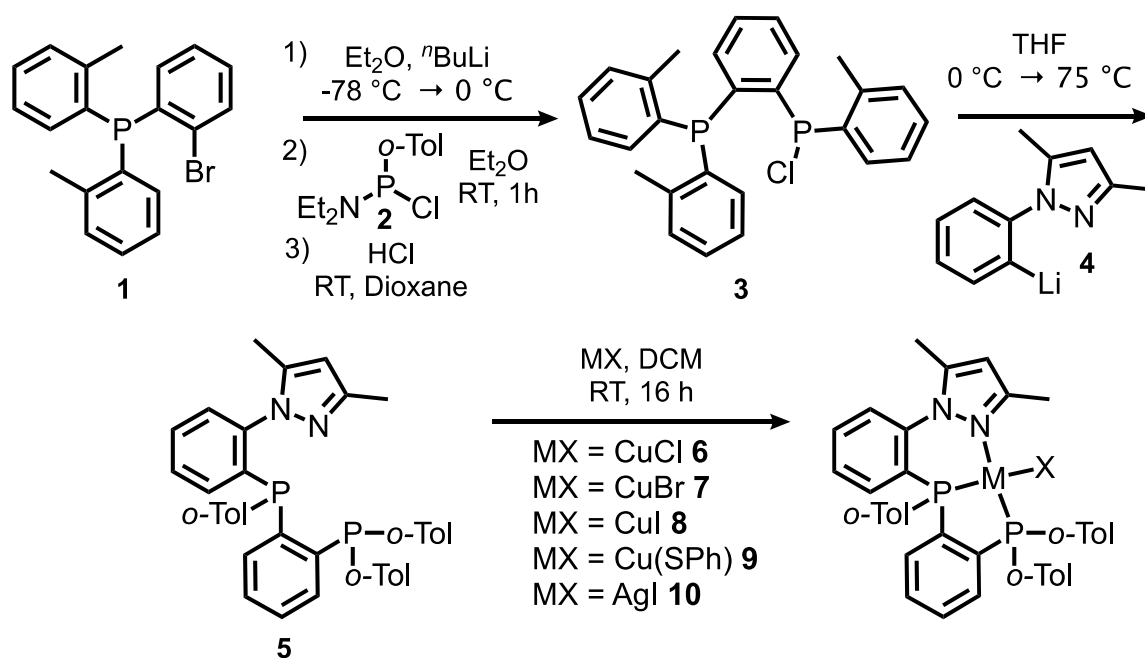
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3 as frequently found. For example, for compound **8** used in an OLED emission layer, a quantum
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5 yield of $\Phi_{\text{PL}} = 63\%$ has been determined. Furthermore, these neutral complexes are, at least in
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7 part, stable and sublimable without thermal decomposition. Hence, using compounds **8** and **9**,
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9 vacuum deposited OLEDs can be fabricated. (See below.)
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12 In the subsequent sections, we first discuss syntheses of the ligand and the complexes and
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14 characterize the new materials chemically as well as by X-ray structure determinations. In the
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16 next section, we present DFT and TD-DFT computational studies to obtain first insight into the
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18 electronic structures of the complexes. Then, we will show that the four compounds studied are
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20 TADF emitters. For a representative example, Cu(dmpzpp)SPh **9**, the TADF and
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22 phosphorescence behavior will be photophysically discussed in detail. Thereafter, we will
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24 present results of vacuum deposited OLEDs using compounds **8** and **9**. Interestingly, the
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26 external quantum efficiency (EQE) of the vacuum deposited device with **8** is as high as 16.4%,
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28 although the Φ_{PL} value in the applied host film amounts only to 63%. Thus, the EQE value
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30 observed is remarkably high. Finally, in a conclusion, strategies for material improvements will
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32 be addressed.
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38 **Syntheses and characterizations**

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41 The new ligand dmpzpp **5** was synthesised in four steps. (2-Bromophenyl)di-*o*-
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43 tolylphosphane **1** and chloro-*N,N*-diethyl-*o*-tolylphosphanamine **2** were prepared as described
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45 in the literature.^{101,102} Key intermediate chloro(2-(di-*o*-tolylphosphanyl)phenyl)(*o*-
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47 tolyl)phosphane **3** was synthesized in a one-pot-reaction following the sequence of a bromine-
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49 lithium exchange at **1**, phosphanylation by amino group protected synthon **2**, finally
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51 deprotecting the intermediate with HCl in dioxane. By this protocol chloro functionalized
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53 phosphorus electrophile **3** could be isolated as colourless solid in 99% yield! **3** was then
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55 converted with 2-(3,5-dimethyl-1*H*-pyrazol-1-yl)phenyl)lithium, generated in situ via an ortho-
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57 directed lithiation to give the dmpzpp ligand **5** as colourless powder in 69% yield.
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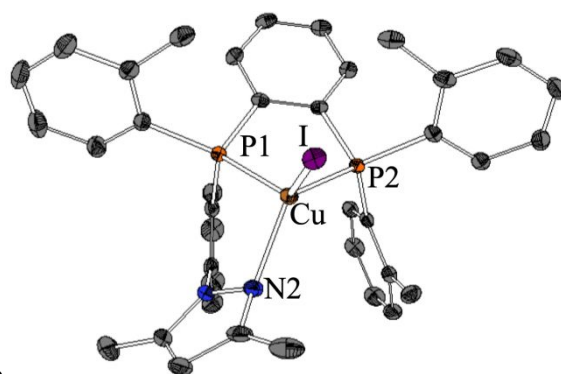
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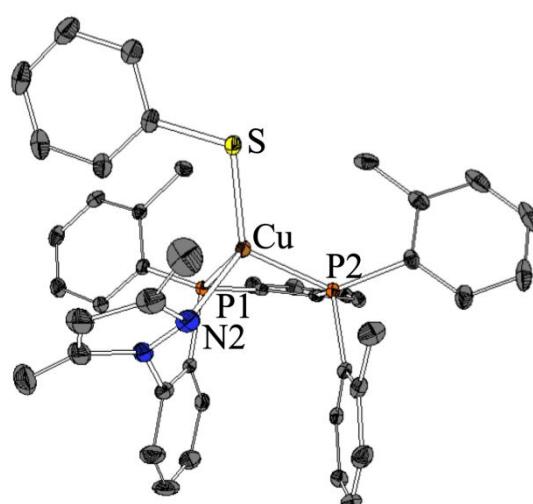
Scheme 1. Synthesis of 3,5-dimethyl-1-(2-((2-(di-*o*-tolyl)phosphanyl)(*o*-tolyl)-phosphanyl)phenyl)-1*H*-pyrazole (dmpzpp **5**) and of highly emissive Cu(dmpzpp)X **6-9** and Ag (dmpzpp)I **10** complexes. The complexes **7** to **10** are characterized below with respect to their emission properties.

In order to synthesize the compounds **6-10**, ligand **5** was dissolved in acetonitrile and reacted with a copper(I) halide (CuCl, CuBr, and CuI), with AgI or with preformed Cu(SPh) at ambient temperature in dichloromethane (DCM). The precipitates formed were collected by centrifugation, washed and dried in vacuum at 60 °C to obtain the desired products **6-9** and **10** as yellow and colorless powders, respectively. (For details see SI-2.) Interestingly, complex **8** can be sublimed at $8.6 \cdot 10^{-7}$ mbar at a temperature of 240-250 °C. The sublimate shows exactly the same NMR spectra as the starting material (see Figure S-14). Thus, this complex is privileged for vacuum processing of an OLED. Compound **9** can also be vacuum processed though at about two orders of magnitude lower vacuum pressure than necessary for **8**. Moreover, all compounds are solution processable without showing high air sensitivity.

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3 Single crystals of the complexes **6-10** could be obtained by layering the corresponding
4 chloroform or acetonitrile solution with *n*-pentane. The complexes crystallize either in a
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6 chloroform or acetonitrile solution with *n*-pentane. The complexes crystallize either in a
7
8 monoclinic (**6,7**) or triclinic (**8-10**) crystal system. Single crystal structures of all dmpzpp Cu(I)
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10 and Ag(I) complexes were determined by X-ray diffraction analyses. Selected bond distances
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12 and angles for **6-10** are listed in Table 1. The crystallographic data and structure refinement
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14 details can be found in the supporting information (see SI-5). As representative examples, the
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16 molecular structures of **8** and **9** are displayed in Figure 1, the others are described in the
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18 supplement (SI-5).
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a



b

Figure 1. Molecular structures of (a) Cu(dmpzpp)I **8** and (b) Cu(dmpzpp)(SPh) **9** from X-ray structure determinations (thermal ellipsoids with 50% probability). Hydrogen atoms and solvent molecules are omitted for clarity.

Table 1. Selected bond distances [Å] and angles [°] based on X-ray diffraction measurements.

Compound	Cu(dmpzpp)C 1 6	Cu(dmpzpp)Br 7	Cu(dmpzpp)I 8	Cu(dmpzpp)(SPh)) 9	Ag(dmpzpp) I 10
M-P1 ^a [Å]	2.2829(7)	2.2508(7)	2.2828(5)	2.2757(4)	2.533(5)
M-P2 [Å]	2.2600(7)	2.2893(7)	2.2756(5)	2.2753(4)	2.488(5)
M-N2 [Å]	2.139(2)	2.148(2)	2.0951(15)	2.2708(16)	2.570(5)
M-X ^b [Å]	2.2674(7)	2.3992(4)	2.5684(2)	2.2403(4)	2.702(6)
P1-M-X [°]	122.45(3)	120.90(2)	123.965(14)	127.314(16)	134.0(3)
P2-M-X [°]	125.26(3)	123.79(2)	115.152(14)	124.065(16)	124.33(16)
N2-M-X [°]	105.50(6)	111.90(6)	112.23(4)	110.22(4)	110.7(2)
P1-M-P2 [°]	87.65(3)	87.66(2)	87.009(17)	87.125(15)	80.83(16)
P1-M-N2 [°]	96.87(6)	93.65(6)	95.81(4)	91.98(4)	84.18(14)
P2-M-N2 [°]	115.59(6)	113.18(6)	119.77(4)	110.77(4)	115.4(2)

a. M = Cu, Ag;

b. X = Cl, Br, I, SPh.

In all complexes, the central metal ion shows a pseudo-tetrahedral configuration, where the P-M-X angles are with 115.152(14)° to 134.0(3)° significantly larger than for an ideal tetrahedron with 109.5°. Therefore, the P1-M-P2 and P1-M-N2 angles with 80.83(16)° to 96.87(6)° are smaller, while the N2-M-X angles with 105.50(6)° to 112.23(4)° and the P2-M-N2 angles with 110.77(4)° to 119.77(4)° fall partly closer to the ideal tetrahedral angle. As expected, the M-X distances increase within the period of the halides and from copper(I) to silver(I) with the effective ionic radii.¹⁰³ The significantly shorter Cu-S distance in the thiophenolato complex **9** is an indicator for the thiophilicity of copper(I), a consequence of a soft acid-base interaction according to Pearson¹⁰⁴ and is even shorter (2.2403(4) Å) than the Cu-S distance in a comparable complex [(Ph₃P)₂(py)CuSH] (2.322(1) Å (py = pyridine)).¹⁰⁵ The Cu-P distances are between 2.2508(7) Å and 2.2829(7) Å being consistent with the ones reported for

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3 [(Ph₃P)₂(py)CuX] with X = Cl, Br, I.¹⁰⁶ The Ag-P distances amount to 2.533(5) Å and
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5 2.488(5) Å for complex **10** and are marginally shorter than the Ag-P distances in
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7 [(Ph₃P)₃AgI].¹⁰⁷
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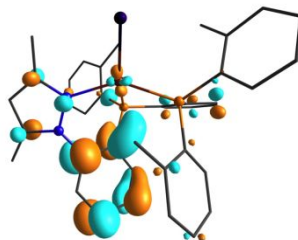
11 12 **Electronic structures. DFT and TD-DFT computations**

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15 Computational studies (PBE-D3(BJ)/def2-TZVPP) were performed on the model compounds
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17 Cu(dmpzpp)I **8** and Cu(dmpzpp)(SPh) **9** to obtain first insights into the electronic structures.
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19 Taking into account that the lowest excited states S₁ and T₁ essentially determine the
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21 luminescence properties of the complexes, the contribution of the HOMO → LUMO transition
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23 was studied. For **8**, this excitation contributes with 99.9% to the S₁ state and with 98.9% to the
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25 T₁ state as calculated for the vertical excitation for the optimized T₁ state geometry. For **9**, the
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27 values are 99.8% and 99.9%, respectively. The Kohn-Sham orbitals of HOMO and LUMO, as
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29 displayed in Figure 2, show only small overlap. This means that the S₁ and T₁ states are of
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31 distinct charge transfer character and it implies relatively small ΔE(S₁-T₁) values and small
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33 transition dipole moments.^{38,49} For Cu(dmpzpp)I **8**, the HOMO is mainly Cu (6.1% d_{x²-y²},
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35 3.1% d_{yz}, 2.8% d_{xz}) and I based and the LUMO is essentially located on the phenyl backbone
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37 of the 2-(3,5-dimethyl-1*H*-pyrazol-1-yl)phenyl fragment. The HOMO of Cu(dmpzpp)(SPh) **9**
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39 is delocalized over Cu (6.1% d_{x²-y²}, 5.8% d_{yz}) and the thiophenolate (S: 18.5% p_z, 10.2% p_x,
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41 Ph: 15.8% p (sum of p-orbitals of the aromatic C atoms)) ligand and the LUMO of **9** is similar
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43 to the one of **8**. For simplicity, we denote the corresponding transitions as metal-to-ligand
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45 charge transfer (MLCT) transitions. Although the calculated transition energies are too small,
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47 if compared to the experimental ones, we assume that the energy differences display the
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49 experimental situation more realistically. Thus, for **8** and **9** we find ΔE(S₁-T₁) = 774 cm⁻¹ (96
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51 meV) and 319 cm⁻¹ (40 meV), respectively, if we refer to the T₁ state optimized geometries.
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60 (Tables S-7 and S-14) The experimental energy gap, only determined for compound **9**, amounts

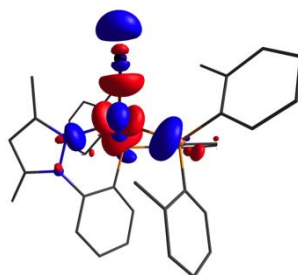
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3 to 600 cm^{-1} (74 meV). (Figure 5, below) These energy separations are sufficiently small, to
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5 predict an occurrence of efficient TADF for these complexes.
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Cu(dmpzpp)I 8

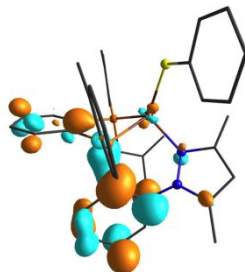


LUMO = -2.51 eV

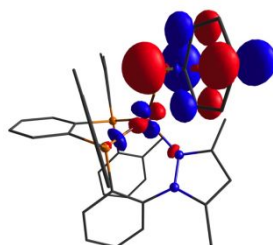


HOMO = -3.36 eV

Cu(dmpzpp)(SPh) 9



LUMO = -2.58 eV



HOMO = -3.37 eV

Figure 2. Kohn-Sham orbitals of Cu(dmpzpp)I **8** and Cu(dmpzpp)SPh **9** calculated for the optimized T_1 state geometry (iso-value = 0.05). Calculations were performed at the PBE-D3(BJ)/def2-TZVPP level of theory. The orbitals shown largely correspond to the NTOs

(natural transition orbitals) of both the T_1 and S_1 states of **8** and **9**, respectively. Thus, HOMO and LUMO match to hole and electron, respectively.

Furthermore, we calculated the geometries of the electronic ground state S_0 , the first excited singlet S_1 and the first excited triplet state T_1 of Cu(dmpzpp)I **8** and Cu(dmpzpp)(SPh) **9**. As expected, the S_0 molecular geometries of **8** and **9** are found to be in reasonable agreement with the geometries obtained from single crystal X-ray diffraction analyses (see SI Tables S-6 and S-13). Usually, for Cu(I) and Ag(I) complexes geometry rearrangements will occur upon excitation. Indeed, when comparing the ground state S_0 geometries with those of the excited states S_1 and T_1 , structural rearrangements are displayed by the calculations, being more distinct for the $S_0 \rightarrow T_1$ than for the $S_0 \rightarrow S_1$ transition. (Tables S-6, S-13, Figures S-20, and S-24) For **8**, the structural rearrangements in the excited state are significantly larger than for **9**. For compound **8**, the I-Cu-P1 angle changes from 121.6° in the S_0 state to 159.0° in the T_1 state or to 125.9° in the S_1 state, which is notably less compared to the T_1 state. Changes of the I-Cu-P2 and I-Cu-N angles are also significant, amounting to 122.8° (S_0)/ 99.4° (T_1) and to 111.0° (S_0)/ 106.7° (T_1), respectively, while the deformation in the S_1 state is almost negligible 122.8° (S_0)/ 122.7° (S_1) and 111.0° (S_0)/ 111.2° (S_1), respectively. For compound **9**, the structural rearrangements are significantly less pronounced, since for this complex, the halide is replaced by the more bulky thiophenolate ligand. The S-Cu-P1 angle changes only from 122.1° (S_0) to 113.2° (T_1) or to 120.1° (S_1), while the angles S-Cu-P2 vary from 123.3° (S_0) to 121.4° (T_1) or 120.3° (S_1) and S-Cu-N from 111.7° (S_0) to 117.0° (T_1) or to 115.7° (S_1). Figure 3 summarizes the geometries in the different states for Cu(dmpzpp)(SPh) **9** to visualize the small geometry changes for the complex with the bulky ligand. From these theoretical predictions, it is expected that the more bulky compound Cu(dmpzpp)(SPh) **9** shows a higher emission quantum yield than Cu(dmpzpp)I **8**. This is consistent with experimental observations (see below).

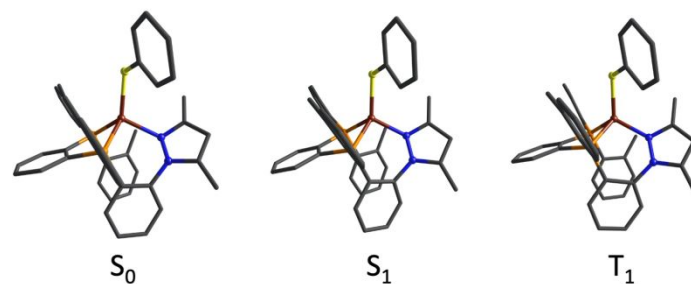


Figure 3. Optimized geometry of Cu(dmpzpp)(SPh) **9** for the S_0 , S_1 , and T_1 states in gas phase. The diagram illustrates that only small distortions occur upon excitation due to the bulky ligands in **9**. Calculations were performed at the PBE-D3(BJ)/def2-TZVPP level of theory. Hydrogen atoms are omitted for clarity.

TADF properties

DFT and TD-DFT calculations show that the lowest excited states are of $^1\text{MLCT}$ (S_1) and $^3\text{MLCT}$ (T_1) character, respectively. For the discussed compounds **8** and **9**, these states stem to more than 98% from HOMO \rightarrow LUMO excitations. These orbitals are essentially localized in the metal-halide or metal-thiophenolato molecular range (HOMO) and on the N,P,P ligand (LUMO), respectively. (Figure 2) Accordingly, a relatively small exchange interaction between the unpaired electrons occurs, giving $\Delta E(S_1-T_1)$ values of several hundred cm^{-1} . Hence, TADF is expected to be found at ambient temperature. Indeed, already simple spectroscopic studies at 300 K and at 77 K can demonstrate this property.^{37,38,49} (Table 2) At low temperature, the compounds exhibit a long-lived $T_1 \rightarrow S_1$ phosphorescence (slow radiative rate). With increasing temperature, the energetically higher lying S_1 state is successively populated opening the additional radiative TADF deactivation path, which leads to a dramatic decrease of the emission decay time or to a drastic increase of the radiative rate. This is particularly distinct, if the 77 K and 300 K decay times of compound **7** are compared, showing a decrease by a factor of more than 360. In parallel, the emission spectra exhibit a slight blue shift (if not hidden in the broad emission bands) due to the higher energy of the S_1 state compared to the T_1 state. (Table 2)

Table 2. Emission data measured for powders. $\lambda(\text{max})$, τ , and Φ_{PL} represent the wavelength of the emission peak maxima, the emission decay time, and the emission quantum efficiency, respectively.

Compound	T/K	$\lambda(\text{max})/\text{nm}$	$\tau/\mu\text{s}$	$\Phi_{\text{PL}}/\%$	k^r/s^{-1} a
7	300	541	9	83	$9.2 \cdot 10^4 \text{ s}^{-1}$
Cu(dmpzpp)Br	77	551	3300	85 ^b	$2.6 \cdot 10^2 \text{ s}^{-1}$
8	300	530	7	82	$12 \cdot 10^4 \text{ s}^{-1}$
Cu(dmpzpp)I	77	530	≈ 420	85 ^b	$20 \cdot 10^2 \text{ s}^{-1}$
9	300	540	5	90	$18 \cdot 10^4 \text{ s}^{-1}$
Cu(dmpzpp)SPh	77	540	680	70	$10 \cdot 10^2 \text{ s}^{-1}$
10	300	479	13	70	$5.4 \cdot 10^4 \text{ s}^{-1}$
Ag(dmpzpp)I	77	484	270/470 ^c	-	-

a. Radiative rate calculated according to $k^r = \Phi_{\text{PL}}/\tau$

b. Estimated value, absolute error $\pm 15\%$

c. Bi-exponential decay.

Although Cu(I) complexes mostly show TADF behavior,^{1,37,38,49} for Ag(I) complexes, TADF is relatively rare.^{48,62,63,64,108,109} This is related to the differences of the oxidation potentials of the 3d/4d-electrons. For Ag^+ it is significantly higher than for Cu^+ .¹¹⁰ Thus, frequently, the 4d-orbitals of Ag^+ complexes are energetically even lower than the occupied orbitals of the organic ligands. Hence, the lowest excited states of the complexes are not of MLCT but of ligand-centered (LC) character and thus, usually show long-lived phosphorescence even at ambient temperature, but no TADF.^{100,111-114} Therefore, the design of Ag(I) complexes that exhibit short-lived TADF requires attention with respect to the choice of the organic ligands. They should induce an energy destabilization of the deep-lying 4d-orbitals. This can be achieved by ligands with good electron donating properties, such as ligands with phosphine coordination to the metal center. Indeed, this is realized in compound **10**. Due to the energy destabilization of the d-orbitals they strongly contribute to the HOMO. As a consequence, TADF emission is observed. Interestingly, as a net result, the transition energy is usually higher than for the related Cu(I) complex with the same ligands. In fact, these properties are displayed in Table 2.

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3 Ag(dmpzpp)I **10** shows blue TADF at λ_{\max} = 479 nm, while Cu(dmpzpp)I **8** exhibits green
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5 emission at 530 nm.
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9 Simple considerations allow us also to qualitatively understand the blue shift of the TADF
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11 emission of Cu(dmpzpp)Br **7** with λ_{\max} = 541 nm to Cu(dmpzpp)I **8** emitting at λ_{\max} = 530 nm.
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13 This shift can be rationalized by a decrease of the ligand field strength from Br to I.¹¹⁵
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15 Accordingly, the resulting smaller splitting of the 3d-orbitals, being involved in the HOMO,
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17 leads to an energy stabilization of the highest-lying 3d-orbital of the iodido complex compared
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19 to the bromido complex, while the LUMO remains almost un-shifted. As a consequence, higher
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21 emission energy occurs for compound **8** than for compound **7**. (Table 2) This trend is similarly
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23 displayed by our TD-DFT computations. (Tables S-4 and S-11) Equivalent behavior has also
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25 been described for other Cu(I) complexes.^{40,52}
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31 In conclusion, all compounds studied exhibit relatively short-lived and efficient TADF at
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33 ambient temperature with high emission quantum yields at radiative TADF decay rates between
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35 $5 \cdot 10^4 \text{ s}^{-1}$ and $18 \cdot 10^4 \text{ s}^{-1}$. At $T = 77 \text{ K}$, one observes only long-lived phosphorescence with
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37 around two orders of magnitude slower decay rates. The emission colors are green to yellow
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39 for the copper complexes and blue for the silver compound.
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43 In the subsequent section, we will present a detailed discussion of phosphorescence and TADF
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45 properties by studying the representative compound Cu(dmpzpp)SPh **9** over the large
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47 temperature range from $T = 1.7 \text{ K}$ to 300 K .
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50 51 **Detailed analysis of phosphorescence and TADF properties of Cu(dmpzpp)SPh 9**

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54 From the series of compounds studied in this contribution, we select Cu(dmpzpp)SPh **9** for
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56 more detailed investigations. It exhibits the highest emission quantum yield of this series of Φ_{PL}
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58 = 90%. This is due to the bulky coordination around Cu and the fast (radiative) TADF decay
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3 rate of $k^r(\text{TADF}) = 18 \cdot 10^4 \text{ s}^{-1}$ ($\tau^r = 5.6 \text{ }\mu\text{s}$). This rate is also relatively fast if compared to other
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5 Cu(I) complexes.^{37,38,49}
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8 *Spectra and quantum yields*

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11 Figure 4 reproduces absorption and emission spectra of Cu(dmpzpp)SPh **9** recorded in various
12 environments. The absorption is measured in dichloromethane (DCM). The different bands
13 recognized in the range of 230 to 350 nm are assigned to ligand centered $\pi\pi^*$ transitions, while
14 the weaker absorptions from about 350 to 450 nm are characterized as MLCT transitions
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16 (compare the computational studies discussed above).
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24 The emission spectra are found with significant Stokes shift. They are very broad as expected
25 for charge transfer (CT) transitions. At ambient temperature, one essentially observes TADF
26 from the S_1 ($^1\text{MLCT}$) state peaking at 540 nm (yellow). The TADF quantum yield of the powder
27 material is with $\Phi_{\text{PL}} = 90\%$ very high. Upon cooling to $T = 77 \text{ K}$, TADF is frozen out and one
28 finds the T_1 ($^3\text{MLCT}$) $\rightarrow S_0$ phosphorescence. The quantum yield drops to $\Phi_{\text{PL}} = 70\%$. Such a
29 behavior is not frequently observed, but compare⁶². It can be rationalized by the drastic decrease
30 of the radiative rate from $k^r(\text{TADF}) = 18 \cdot 10^4 \text{ s}^{-1}$ (300 K) to $k^r(\text{phos}) = 10 \cdot 10^2 \text{ s}^{-1}$ (77 K) by
31 more than two orders of magnitude. (Table 2) Using the relation for the non-radiative rate $k^{\text{nr}} =$
32 $(1 - \Phi_{\text{PL}})/\tau$, $k^{\text{nr}}(\text{phos}) = 4.4 \cdot 10^2 \text{ s}^{-1}$ (77 K) is obtained. Thus at $T = 77 \text{ K}$, the non-radiative
33 processes can moderately compete with the radiative one. On the other hand, at $T = 300 \text{ K}$ with
34 $k^{\text{nr}}(\text{TADF}) = 2 \cdot 10^4 \text{ s}^{-1}$ (300 K) the radiative rate strongly prevails the non-radiative one.
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51 With temperature decrease from $T = 300 \text{ K}$ to 77 or to 50 K, when TADF is frozen out, one
52 would expect a small red shift due to the $\Delta E(S_1-T_1)$ energy gap. However, if the gap is relatively
53 small (for **9**: 600 cm^{-1} , 74 meV, see below), the shift may be hidden by the broad band emission.
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55 Such a behavior has also been observed for other TADT Cu(I) complexes.^{52,55}
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The emission of compound **9** in less rigid environments than in crystalline powders shows significant reduction of the quantum yield. For the compound doped in relatively soft PMMA (poly(methyl methacrylate)) and dissolved in fluid DCM, respectively, Φ_{PL} values of 56% and only $\approx 1\%$ are observed. (Table 3, below) This Φ_{PL} reduction is explained by the more distinct geometry changes upon excitation in the softer environment. As a consequence, the non-radiative deactivation processes can become more pronounced due to the higher FC factors of the excited and ground state vibrational wavefunctions. Thus, the quantum yield drops as compared to a more rigid environment. Moreover, the geometry change usually results in a red shift of the emission spectra with respect to the more rigid crystalline environment. Indeed, a red shift of ≈ 4 nm is found for the peak maxima. (Figure 4)

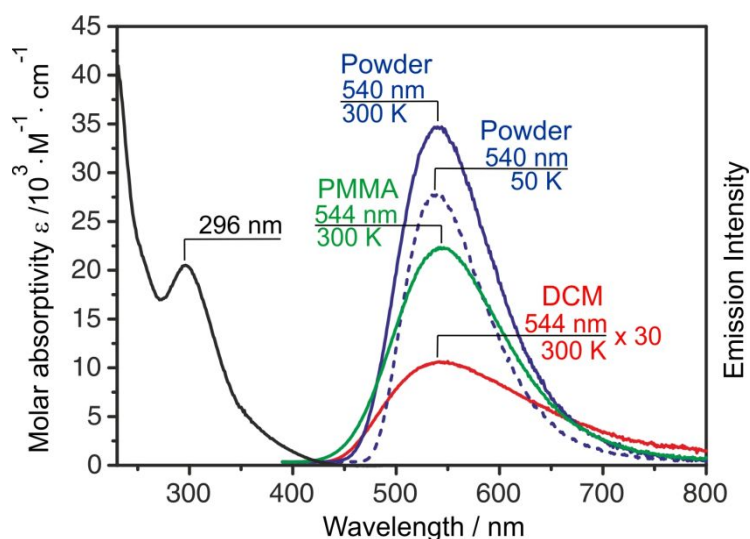


Figure 4. Absorption and emission spectra of Cu(dmpzpp)SPh **9**. Absorption spectrum (black line) recorded in dichloromethane (DCM) solution (concentration $c \approx 10^{-5}$ M) at 300 K, emission spectra as powder (blue lines), doped in PMMA (green line, $c \approx 1$ w%), and dissolved in DCM solution (red line, $c \approx 10^{-5}$ M). The emission intensity of the solution spectrum is displayed 30 times enlarged. $\lambda_{\text{exc}} = 330$ nm.

For completeness, we want to point out that an occurrence of a geometry distortion has an important consequence.^{11,30,38,58} Typically, it is not possible to conclude on emission properties of individual molecules from investigations of neat materials (powders), because inter-

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3 molecular effects, such as resonant energy transfer to neighboring molecules and finally to
4 quenching impurities, will usually at least influence the emission decay behavior or quantum
5 yield and frequently even quench the emission totally (concentration quenching). However, for
6 most Cu(I) complexes, the geometry distortions upon excitation are strong enough, even in the
7 neat powder phase, to lower the excited states energies sufficiently, so that resonant energy
8 transfer to adjacent non-excited molecules is largely prevented. Accordingly, the excitation is
9 trapped at the initially excited molecule. Such a self-trapping effect^{116,117} is the basis for the
10 studies presented in this contribution.
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22 The explanation given above is nicely supported by a recent investigation with Cu(I) tripod
23 complexes with a specifically designed rigid and bulky ligand coordination, in particular,
24 [Cu(tpym)(P(*o*-butyl-ph)₃)]PF₆, wherein tpym represents tris(2-pyridyl)methane and P(*o*-butyl-
25 ph)₃ tris(*ortho*-*n*-butylphenyl)phosphine.⁵⁸ Accordingly, geometry distortions in the excited
26 state are largely prevented. As a consequence, a TADF material was developed that shows the
27 highest emission quantum yield in fluid solution reported so far (86%).⁵⁸ Interestingly, the
28 corresponding powder material displays a significantly smaller quantum yield (56%).
29 Obviously, the self-trapping effect is partly suppressed in the rigidified molecule. Thus,
30 concentration quenching can occur, in contrast to the situation found for almost all other Cu(I)
31 complexes.
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46 In conclusion, the emission measured at ambient and lower temperature, even if cooled to T =
47 1.7 K (not reproduced), show only broad bands. Accordingly, investigation of such spectra does
48 not disclose detailed information of the compound's photophysical properties. However, from
49 studies of the emission decay behavior, we know from the data presented in Table 2 that
50 significant changes occur with temperature change. Therefore, in the subsequent section, we
51 will study the decay behavior. Indeed, it will be shown that a deeper insight in the electronic
52 structure and radiative relaxation properties thus becomes possible.
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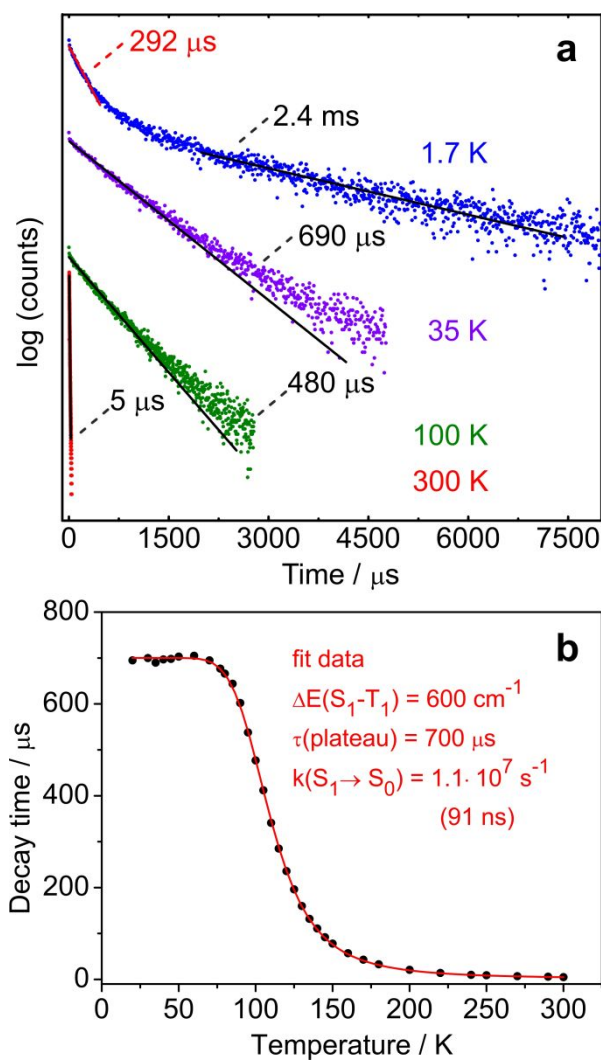


Figure 5. Emission decay behavior of Cu(dmpzpp)SPh **9** powder. (a) reproduces emission decay curves measured at selected temperatures. Approximated decay components (fits) are inserted. (b) shows the emission decay time versus temperature (black dots) and the fit according to eq. (1) (red line). $\lambda_{\text{exc}} = 378 \text{ nm}$, excitation pulse width $< 100 \text{ ps}$, $\lambda_{\text{det}} = 530 \text{ nm}$.

Emission decay, energy levels, and transition rates

Figure 5a reproduces decay curves for several selected temperatures. Apart from the 1.7 K decay that follows nearly bi-exponential decay dynamics (for details see below), we can approximate the decay at higher temperature mono-exponentially. Such a behavior corresponds to fast equilibration between the involved emitting states, i.e. faster than the decays of the

individual states, and it excludes extensive inhomogeneity effects within the ensemble of emitting complexes (but compare ref.¹¹⁸). In this situation, we can apply a Boltzmann-type equation to characterize the temperature dependence of the emission decay time $\tau(T)$ according to^{1,11,49,119}

$$\tau(T) = \frac{3 + \exp\left(-\frac{\Delta E(S_1-T_1)}{k_B T}\right)}{3k(T_1) + k(S_1)\exp\left(-\frac{\Delta E(S_1-T_1)}{k_B T}\right)} \quad (1)$$

Herein, $\Delta E(S_1-T_1)$ refers to the energy separation between the S_1 and T_1 states. $k(S_1)$ and $k(T_1)$ are the rates of the $S_1 \rightarrow S_0$ and $T_1 \rightarrow S_0$ transition, respectively. k_B is the Boltzmann constant. If we assume coarsely constant rates and quantum yields over the whole temperature range, fitting of eq. (1) to the experimental data given in Figure 5b leads to a well matching fit curve. This dependence displays nicely the changes of the emission processes within the temperature range studied. At low temperature between $20 \leq T \leq 70$ K, one observes a plateau showing the decay time of $\tau(\text{phos}) = 700 \mu\text{s}$, displaying the phosphorescence of compound **9**. It corresponds to the radiative decay time of $\tau^r(\text{phos}) = \tau(\text{phos})/\Phi_{\text{PL}} = 1\text{ms}$. Above $T = 70$ K, the decay time sharply decreases and finally reaches a value of $5 \mu\text{s}$ at ambient temperature. The decrease is induced by thermal population of the higher lying S_1 state and by the fact that the spin-allowed $S_1 \rightarrow S_0$ fluorescence is drastically faster than the spin-forbidden $T_1 \rightarrow S_0$ phosphorescence, leading to the TADF effect.

For completeness, it is remarked that a prompt $S_1 \rightarrow S_0$ fluorescence with a decay time of 91 ns (Figure 5b and below) is not observed with our experimental time resolution, because the competing $S_1 \rightarrow T_1$ ISC process, occurring in about 30 ps ¹²⁰, is by more than three orders of magnitude faster.

Using the fit procedure according to eq (1), we obtain the energy gap of $\Delta E(S_1-T_1) = 600 \text{ cm}^{-1}$ (74 meV) and the rate for the $S_1 \rightarrow S_0$ transition of $k(S_1) = 1.1 \cdot 10^7 \text{ s}^{-1}$ (91 ns). (Figures 5 and 6, Table 3) Both parameters are of dominating importance for the TADF decay time at ambient temperature. The resulting value of $\tau(\text{TADF}) = 5 \text{ }\mu\text{s}$ or of $\tau^r(\text{TADF}) = 5.6 \text{ }\mu\text{s}$ is relatively short if compared to conventional TADF emitters. Usually, short decay times are required for obtaining long-lived OLED devices.^{1,95}

Table 3. Photophysical data of Cu(dmpzpp)SPh **9** as powder, doped in PMMA ($\approx 1 \text{ w}\%$), and in fluid dichloromethane (DCM) solution ($c \approx 10^{-5} \text{ M}$).

	powder	PMMA	DCM
λ_{max} (300 K)	540 nm	544 nm	544 nm
Φ_{PL} (300 K)	90%	56%	1%
τ (300 K)	5 μs	7 μs	4 μs
k^r (300 K)	$18 \cdot 10^4 \text{ s}^{-1}$	$8.0 \cdot 10^4 \text{ s}^{-1}$	$0.25 \cdot 10^4 \text{ s}^{-1}$
k^{nr} (300 K) ^a	$2.0 \cdot 10^4 \text{ s}^{-1}$	$6.3 \cdot 10^4 \text{ s}^{-1}$	$25 \cdot 10^4 \text{ s}^{-1}$
λ_{max} (50 K)	540 nm		
Φ_{PL} (77 K)	70%		
$\tau(T_1, 50 \text{ K})$ plateau	700 μs		
$k^r(T_1, 50 \text{ K})^b$	$10 \cdot 10^2 \text{ s}^{-1}$		
$k^{\text{nr}}(T_1, 50 \text{ K})^{\text{a, b}}$	$4.3 \cdot 10^2 \text{ s}^{-1}$		
$k^r(S_1 \rightarrow S_0)^c$	$1.1 \cdot 10^7 \text{ s}^{-1}$ (91 ns)		
$\Delta E(S_1-T_1)^c$	600 cm^{-1}		
$\tau(\text{I}), \tau(\text{II}),$ $\tau(\text{III})^d$	2.4 ms, 2.25 ms, 292 μs		

a. Non-radiative rate calculated from $k^{\text{nr}} = (1-\Phi_{\text{PL}})/\tau$

b. It is assumed that $\Phi_{\text{PL}}(50 \text{ K}) \approx \Phi_{\text{PL}}(77 \text{ K})$

c. From a fit of the $\tau(T)$ plot (Figure 5b) using eq. (1)

d. Individual decay times of the three triplet substates, determined at $T = 1.7 \text{ K}$

The decay behavior at very low temperature, at $T = 1.7$ K for example, is distinctly non-monoeponential. This is a consequence of slow thermalization between the three T_1 substates I, II, and III according to slow rates of spin-lattice relaxation (SLR) at low temperature, if the zero-field splitting $\Delta E(\text{ZFS})$ between these states is small, e. g. less than a few cm^{-1} .¹²¹ In this situation, the individual triplet substates emit independently with their intrinsic decay times $\tau(\text{I})$, $\tau(\text{II})$, and $\tau(\text{III})$. From the decay curve measured at $T = 1.7$ K, we identify two decay components of 292 μs and 2400 μs , but the third one is hidden. (Figure 5a) However, at higher temperature, at 10 or 20 K for example, SLR rates become fast¹²¹ and an average decay time τ_{av} of the three substates is obtained, which can be expressed by^{121,122,123}

$$\tau_{\text{av}}(T_1-S_0) = 3(\tau(\text{I})^{-1} + \tau(\text{II})^{-1} + \tau(\text{III})^{-1})^{-1} \quad (2)$$

Using this equation, we can estimate the hidden component. If τ_{av} is given by the experimental phosphorescence decay time of 700 μs (plateau) and two substate decay components are the ones displayed in Figure 5a in the $T = 1.7$ K decay, we can estimate the hidden component to ≈ 2.25 ms. This value is not very different from the longer-lived component used for this estimate. Two long-lived and similar decay components have also been reported for other Cu(I) compounds.^{30,38,49,52} To summarize, the individual decay times of the substates of compound **9** are $\tau(\text{I}) = 2.4$ ms, $\tau(\text{II}) = 2.25$ ms, and $\tau(\text{III}) = 292$ μs without having the information on the energetic order of the three substates. Table 3 and Figure 6 summarize the emission data worked out for Cu(dmpzpp)SPh **9**.

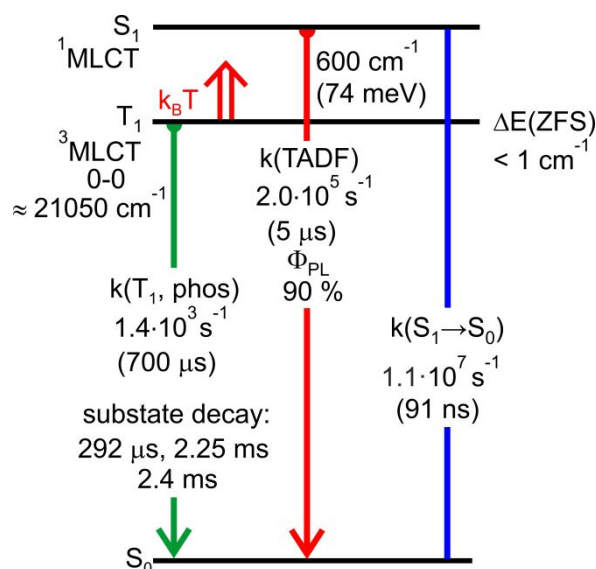


Figure 6. Simplified energy level diagram and decay data for Cu(dmpzpp)SPh **9** powder. The T_1 state consists of three triplet substates I, II, and III that emit independently at $T = 1.7\text{ K}$ with estimated values of $\tau(\text{I}) = 2.4\text{ ms}$, $\tau(\text{II}) = 2.25\text{ ms}$, and $\tau(\text{III}) = 292\text{ }\mu\text{s}$. The 0-0 energy given for the $T_1 \rightarrow S_0$ transition is estimated from the blue energy flank of the phosphorescence band as displayed in Figure 4. The values given for $k(T_1, \text{phos})$ and $k(\text{TADF})$ refer to a representative temperature of the $\tau(T)$ plateau (for example $T = 50\text{ K}$) and 300 K , respectively.

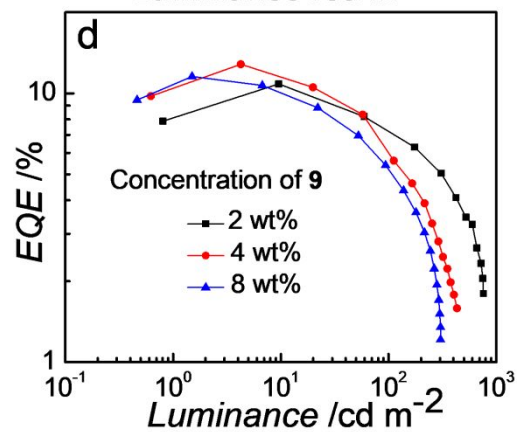
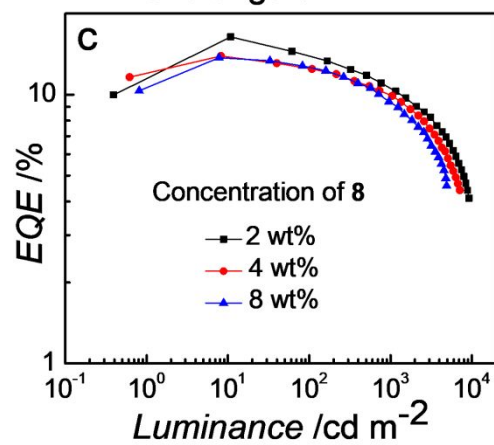
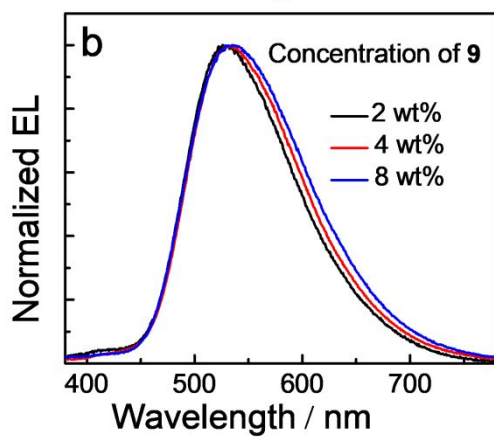
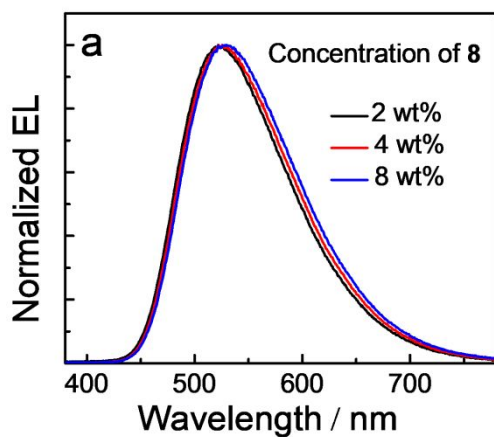
From the decay behavior at $T = 1.7\text{ K}$, we can conclude¹²¹⁻¹²⁵ on a $\Delta E(\text{ZFS})$ value of $< 1\text{ cm}^{-1}$ (0.1 meV), presumably $\ll 1\text{ cm}^{-1}$.¹²⁵ Such a small ZFS combined with the very long phosphorescence decay time of $\tau(\text{phos}) = 700\text{ }\mu\text{s}$ ($\tau^r(\text{phos}) = 1\text{ ms}$) demonstrates¹¹ very weak spin-orbit coupling (SOC) of higher lying singlets to at least to one of the emitting T_1 substates. This can be understood taking the following quantum mechanical consideration into account: SOC between the S_1 and T_1 states that stem from the same HOMO \rightarrow LUMO excitation is almost negligible^{28,32,38,49} (El-Sayed rule¹²⁶). Among the series of compounds discussed, Cu(dmpzpp)Br **7** displays the El-Sayed forbiddenness particularly clearly: Although, $\Delta E(S_1 - T_1)$ amounts only to 680 cm^{-1} (84 meV) (Figure S-27) SOC between the S_1 and T_1 states seems to be unimportant, since the phosphorescence decay time $\tau^r(\text{phos})$ at the plateau is as long as 4.0 ms . (Figure S-27) Obviously, the El-Sayed rule, originally presented for purely organic

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3 molecules, is also valid for organo-transition metal compounds. On the other hand, efficient
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5 SOC requires quantum mechanical admixture(s) with states that result from a different orbital
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7 configuration involving a different d-orbital than the one contributing to the HOMO, for
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9 example, HOMO-1. According to results from TD-DFT calculations of compound **9**, the next
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11 nearest singlet state with a different d-orbital contribution stems from the HOMO-1→LUMO
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13 excitation, giving the state S_5 . However, the energy separation that would be of crucial
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15 importance for quantum mechanical admixtures^{28,127,128}, in this case $\Delta E(S_5-T_1)$, is as large as 0.89
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17 eV (7180 cm^{-1}). (Table S-17). Usually, at such a big energy separation direct SOC is not
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19 efficient.³⁸ Interestingly, the situation for Cu(dmpzpp)Br **7** and Cu(dmpzpp)I **8** concerning the
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21 energy separations to admixing singlet states is similar (Tables S-5, S-10), but for
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23 Cu(dmpzpp)Br **7** the phosphorescence decay time is even about five times longer than for **9**.
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25 (Table 2) Hence, it is expected that for all three compounds other processes than direct SOC,
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27 such as spin-vibronic coupling^{129,130,131}, have to be considered that open the phosphorescent
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29 decay paths. In contrast, the TADF process is based, after thermal activation, on a spin-allowed
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31 singlet-singlet emission and thus, is more than two orders of magnitude faster than the
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33 phosphorescent decay.

40 41 **Electroluminescent properties of Cu(dmpzpp)I **8** and Cu(dmpzpp)SPh **9****

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43 To evaluate the electroluminescent (EL) properties of neutral and sublimable Cu(I) complexes
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45 **8** and **9**, vacuum-deposited OLEDs with the device structure of ITO/HAT-CN(5 nm)/TAPC (40
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47 nm)/TCTA (10 nm)/TCTA: DPEPO: Cu complex (10 nm)/DPEPO (40 nm)/LiF (1.2 nm)/ Al
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49 (100 nm) were fabricated and characterized. In these devices, HAT-CN (1,4,5,8,9,11-
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51 hexaazatriphenylene hexacarbonitrile) was used as hole-injecting layer (HIL), TAPC (1,1-bis-
52
53 (4-bis(4-methylphenyl)-amino-phenyl)-cyclohexane) as hole-transporting layer, TCTA
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55 (4,4',4''-tris(*N*-carbazolyl)-triphenylamine) as electron/exciton-blocking layer, and DPEPO
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57 (bis[2-(diphenylphosphino)phenyl] ether oxide) as electron-transporting layer. The mixture of
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3 TCTA and DPEPO with a weight ratio of 1:1 was used as the co-host of the emitting layer
4 (EML) to maximize the performances of our Cu(I) complex based OLEDs. The chemical
5 structures of HAT-CN, TAPC, TCTA and DPEPO are shown in the SI (Scheme 1). Compounds
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10 **8** and **9**, respectively, were used as the emitting dopants with concentrations of 2, 4, and 8 wt%
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12 in the EML.
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3 **Figure 7. a, b** Normalized electro-luminescence spectra at 4 V and **c, d** external quantum
4 efficiency (EQE) *versus* luminance characteristics of vacuum-deposited OLEDs with
5 Cu(dmpzpp)I **8** and Cu(dmpzpp)SPh **9** as emitting dopants with concentrations of 2, 4, and
6 8 wt%, respectively. Note the different luminance scales of **c** and **d**.
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14 Normalized EL spectra and EQE versus luminance characteristics of these devices are depicted
15 in Figure 7, while the luminance and current density, respectively, versus voltage curves are
16 displayed in the SI (Figures S-28 and S-29). Key performances of these devices are summarized
17 in Table 4. As shown in the Figures 7a and 7b, EL spectra of both compounds **8** and **9** became
18 broader with concentration increase. Similar phenomena were also found in the photo-
19 luminescent (PL) spectra of **8** as depicted in Figure S-30 probably being the result of increasing
20 inhomogeneity of the dopants in the host. Peak maxima of the EL spectra of compounds **8** and
21 **9** are located at around 521 and 535 nm, respectively. This is consistent with the corresponding
22 PL spectra measured in thin TCTA-DPEPO films. Thus, it is concluded that the OLED emission
23 represents TADF of the Cu(I) complex dopants. (The values given for the emission peak
24 maxima slightly deviate from those measured for the powder materials (Table 2) due to the
25 different polarities and rigidities of the emitter's environments.) It is worth to note that the PL
26 spectra of **8** and **9** are quite different. Only one single PL band is seen for compound **8** doped
27 in the TCTA-DPEPO host, while two PL bands are found for the host doped with **9**. (Figure S-
28 30) These bands result from compound **9** and from the host material. Therefore, it is indicated
29 that energy transfer from the host to the Cu(I) compound **9** is inefficient. In fact, a weak host
30 emission can also be observed in the EL spectra of the OLED doped with **9**. The host emission
31 decreases with increasing concentration due to more efficient energy transfer. Nevertheless, the
32 relative intensity of the host emission in the EL spectra is much lower than in the PL spectra,
33 indicating that charge-trapping plays an important role in the EL process of the device with **9**.
34 For charge-trapping OLEDs, excitons directly form and recombine on the emitting dopants
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3 without using the step of energy transfer from the electrically excited molecules of the host to
4 the emitter.¹³² Compared to the devices based on energy transfer, the current density versus
5 voltage characteristics of charge-trapping devices strongly depend on the dopant concentration
6 because charge-trapping on emitting dopants decreases the charge carrier mobility in the
7 EML.^{132,133} As shown in Figure S-29, the decrease of current density with increasing
8 concentration at a given driving voltage supports the charge-trapping mechanism of the **9**
9 devices. With the green light emitting OLED based on compound **8** with a doping concentration
10 of 2 wt%, a maximum external quantum efficiency of EQE = 16.4% and a maximum luminance
11 of 9260 cd m⁻² in the measured voltage range up to 7.5 V at CIE coordinates of (0.33; 0.52)
12 were achieved. (Figure 7 c) If we take into account that the PL quantum yield of compound **8**
13 doped in the TCA-DPEO host amounts only to 63% the EQE value obtained is quite
14 remarkable, because for an extrapolated/assumed value of $\Phi_{\text{PL}}=100\%$, one could expect to
15 achieve an EQE of 26%. This would be a very high value for a conventional OLED device and
16 thus, shows that the device structure is very well chosen.^{1,4,8} Presumably, there is still potential
17 for Φ_{PL} increase by further optimizing the complex' chemical structure (see below). To
18 summarize, the efficiency of this device is among that of the best OLEDs fabricated with Cu(I)
19 compounds by vacuum-deposition technique, so far.^{16,31,34,35,39,48,56,60,134-136}

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44 Since the Φ_{PL} value of compound **9** doped in a thin PMMA film with 56% is lower than Φ_{PL} of
45 **8** with 65% in PMMA and since the energy transfer from the host to compound **9** is inefficient,
46 the maximum EQE of device with compound **9** does not reach the value found for the OLED
47 based on compound **8**. It is notable that with increasing luminance, the efficiency roll-off of the
48 device with **9** is significantly more expressed than for the device with **8**. (Figures 7 c and d)
49 Moreover, the maximum luminance obtained for the device with **9** is less than 1000 cd m⁻². To
50 find an explanation for these pronounced effects, EL spectra of the device with **9** with 4 wt%
51 dopant concentration were investigated at various driving voltages from 4 to 12 V. (Figure S-

31) With increasing voltage, a new emission band with a maximum at about 565 nm appeared and increased together with an increase of the host emission as shown in Figure S-31b. Accordingly, it is indicated that compound **9** may not be stable when high electric fields are applied and could account for the pronounced efficiency roll-off and the low maximum luminance of OLEDs based on compound **9**. Possibly, the CuS-C bond of the thiophenylate ligand breaks and a stable CuS coordination results. For comparison, as depicted in Figure S-31a, the EL spectra of the device with compound **8** at 4 wt% dopant concentration stayed unchanged with increasing driving voltage. Obviously, the device using Cu(dmpzpp)I **8** is more stable even at a luminance of almost 10000 cd m⁻².

Table 4. Key performances of OLEDs with Cu(dmpzpp)I **8** and Cu(dmpzpp)SPh **9** emitters.

Cu(I) complex (concentration)	L_{max} [cd m ⁻²] ^a	CE		PE		EQE		CIE [(x, y)] ^e
		max	at 1000 cd m ⁻²	max	at 1000 cd m ⁻²	max	at 1000 cd m ⁻²	
8 (2 wt%)	9260	41.9	29.1	43.8	21.3	16.4	10.2	0.33, 0.52
8 (4 wt%)	8080	39.5	28.4	41.4	20.3	13.9	9.88	0.35, 0.52
8 (8 wt%)	7600	38.7	26.8	40.5	18.3	13.7	9.38	0.35, 0.52
9 (2 wt%)	760	30.5	N/A	28.2	N/A	10.8	N/A	0.35, 0.51
9 (4 wt%)	433	36.5	N/A	31.8	N/A	12.8	N/A	0.36, 0.52
9 (8 wt%)	304	33.5	N/A	29.2	N/A	11.6	N/A	0.37, 0.52

a. Maximum luminance; b. current efficiency; c. power efficiency; d. external quantum efficiency e. CIE color coordinates at 100 cd m⁻².

Conclusion and design considerations for future developments

In this report, we present a new type of Cu(I) and Ag(I) complexes using a rigid tri-dentate N,P,P-ligand, dmpzpp, with the chemical structure shown in Scheme 1. As additional mono-dentate ligand, we apply charged Cl⁻, Br⁻, I⁻, and SPh⁻. Hence, the complexes are neutral and sublimable. The powder materials of all complexes exhibit bright TADF of up to 90% photo-

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3 luminescence quantum yield at relatively short emission decay time or fast decay rate at ambient
4 temperature. For example, for Cu(dmpzpp)SPh **9**, the radiative rate amounts to $k^r(\text{TADF}) =$
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6 $1.8 \cdot 10^5 \text{ s}^{-1}$ (5.6 μs). (Table 3) This rate is by a factor of almost 200 faster than the
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8 phosphorescence radiative rate of $k^r(\text{phos}) = 1.0 \cdot 10^3 \text{ s}^{-1}$ (1 ms). Accordingly, the TADF process
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10 opens a very efficient radiative decay path, which strongly predominates non-radiative
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12 quenching. Thus, after freezing out the TADF decay path, for example, by cooling to $T = 77 \text{ K}$,
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14 Φ_{PL} drops. Therefore, short radiative decay time at ambient temperature is essential for
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16 obtaining high emission quantum yields. Maybe, applying a design strategy that leads to an
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18 increase of the $S_1 \rightarrow S_0$ transition rate (see eq.(1)), might result in an even further increase of
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20 Φ_{PL} . In a recent study with a similar challenge, we could successfully show that such a rate
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22 increase was achievable for Ag(I) complexes by changing the relative orientation of HOMO
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24 and LUMO.^{49,62} This strategy allowed us to distinctly increase the rate $k(S_1-S_0)$ giving
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26 $\Phi_{\text{PL}}(\text{TADF}) = 100\%$.

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34 The compounds studied cover a large emission color range from yellow to green and to blue
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36 for Cu(I) and Ag(I) complexes, respectively, in all cases at high emission quantum yield. Maybe
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38 even red emission at high Φ_{PL} might be realizable with this class of compounds. Finding
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40 efficient red-light emitters is still a challenge due to quenching processes according to the
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42 energy gap law.⁹⁹ For our materials, for example, substitution of the N,P,P-ligand with electron-
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44 withdrawing groups might lead to such a red shift. This concept, however, is only suitable, if
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46 the required fast TADF rate $k^r(\text{TADF})$ can be maintained.

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51 The class of materials studied is attractive for testing OLED properties. In fact, using well
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53 sublimable Cu(dmpzpp)I **8**, we find a remarkable OLED performance with an external quantum
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55 efficiency of $\text{EQE} = 16.4\%$, although compound **8** being doped with 2% in an emission layer
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57 (TCTA : DPEPO co-host with a 1 : 1 ratio) exhibits a photo-luminescence quantum yield of
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59 only $\Phi_{\text{PL}} = 63\%$, significantly lower than of the powder material. If we formally assume $\Phi_{\text{PL}} =$
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3 100% for the emitter complex in this host, we would expect to find an EQE value of 26%, which
4 demonstrates very good OLED performance. It corresponds to the maximum of out-coupling
5 of light generated by electro-luminescence in conventional OLED device structures, if 100%
6 internal efficiency is obtained.^{1-4,9} Obviously, the device structure developed is well suited for
7 compound **8**. Therefore, future developments for increasing EQE should try to design
8 compounds that exhibit almost 100% photo-luminescence quantum yield. However, occurrence
9 of high Φ_{PL} is not only related to the radiative rates as discussed above, but also to non-radiative
10 processes. These are connected to the rigidity of the molecular structure. For example, the
11 structure with bulky ligands strongly reduces distortions that usually occur in Cu(I) or Ag(I)
12 complexes after MLCT excitation. The less distinct such distortions are, the less significant is
13 non-radiative deactivation and the higher is the emission quantum yield.^{38,99} However, the
14 extent of the geometry change is usually not only given by the intrinsic molecular rigidity. In
15 most cases, also the environment or the host rigidity is highly important.³⁸ Even the compound
16 with the most bulky ligand structure studied in this contribution, Cu(dmpzpp)SPh **9**, exhibits a
17 drastic decrease of Φ_{PL} from 90% as rigid powder, to 56% in PMMA, and finally to only 1% in
18 fluid DCM solution. Obviously, this trend indicates that the molecular rigidity of these
19 compounds requires an additional external or matrix-induced component to reach high Φ_{PL}
20 values. However, the host materials usually available for OLED emission layers do not seem to
21 provide sufficient external rigidity. Hence, one is faced with the challenge to design an even
22 more rigid ligand structure. This could already be demonstrated⁵⁸, though for a charged and
23 thus, not well sublimable compound. Maybe in future, a rigid tetra-dentate ligand coordination
24 might be realizable. However, if so, it is equally important not to lose TADF efficiency.

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In summary, the new compounds presented do not only exhibit attractive TADF properties and
OLED performance, they also open promising options for developments of future materials.

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14 15 **Associated Content**

16 17 **Supporting Information**

18 The Supporting Information is available free of charge on the ACS Publications website at
19 DOI:...

20 Syntheses, NMR-Data, sublimation details of **8** and **9**, X-ray crystallography data, quantum
21 chemical calculations, photophysical characterization of **7**, and electro-luminescence properties
22 of OLEDs with **8** and **9** emitters. (PDF)
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30 31 **Accession Codes**

32 CCDC 1958853 (**6**), 1958854 (**7**), 1958855 (**8**), 1958861 (**9**), and 1958859 (**10**) contain
33 supplementary crystallographic data for this paper. This information can be obtained free of
34 charge via www.ccdc.cam.ac.uk/data_request/cif or by e-mailing
35 data_request@ccdc.cam.ac.uk or by contacting The Cambridge Crystallographic Data Centre,
36 12 Union Road, Cambridge CB2 1EZ, U.K.; fax: + 44 1223 336033.
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42 43 **Note**

44 The authors declare no competing financial interest.
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52 53 **References**

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