Title: Printed miniature robotic actuators with curvature-induced stiffness control inspired by
the insect wing

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9 Abstract: Stimuli-responsive actuating materials offer a promising way to power insect-scale 10 robots, but a vast majority of these material systems are too soft for load bearing in different 11 applications. While strategies for active stiffness control have been developed for humanoid-12 scale robots, for insect-scale counterparts for which compactness and functional complexity are 13 essential requirements, these strategies are too bulky to be applicable. Here, we introduce a 14 method whereby the same actuating material serves not only as the artificial muscles to power 15 an insect-scale robot for load bearing, but also to increase the robot stiffness on-demand, by 16 bending it to increase the second moment of area. This concept is biomimetically inspired by 17 how insect wings stiffen themselves, and is realized here with manganese dioxide (MnO₂) as a 18 high-performing electrochemical actuating material printed on metallized polycarbonate films 19 as the robot bodies. Using an open-electrodeposition printing method, the robots can be rapidly 20 fabricated in one single step in ~15 minutes, and they can be electrochemically actuated by a potential of ~1 V to produce large bending of ~500° in less than 5 s. With the stiffness 21 22 enhancement method, fast (~5 s) and reversible stiffness tuning with a theoretical increment by 23 ~4000 times is achieved in a micro-robotic arm at ultra-low potential input of ~ 1 V, resulting 24 in an improvement in load-bearing capability by about 4 times from $\sim 10 \,\mu\text{N}$ to $\sim 41 \,\mu\text{N}$.

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1 1. Introduction

2 Recently, there has been keen interest in using stimuli-responsive actuating materials to power robots that are compact and bio-mimicking [1, 2, 3]. In this paradigm, a robot, often in 3 the insect scale, can produce lively motions with complex functions via strategically positioning 4 5 an actuating material at specific locations of the robot body [4, 5]. A popular strategy is by 3D 6 printing of actuating materials [6, 7] – a research topic now known as "4D printing" – with 7 examples including shape memory polymers and composite hydrogels [8]. Recently, transition-8 metal oxides have been developed as high-performing multi-stimuli driven actuating materials 9 [2, 3, 9, 10, 11] and methods have been developed to print them "on-demand" onto soft substrates to produce intricate insect-scale robots [12]. Such systems are lightweight and have 10 11 high flexibility for motions, but a drawback is that they have low stiffness that decreases the 12 load-bearing capability and therefore applicability.

13 For larger robotic devices, some strategies have been proposed to tune the stiffness of soft 14 robots, although the applicability of these methods to insect-scale robots requiring high 15 compactness and complex functionality is limited. Humanoid-scale robots can be stiffened by 16 means of using granular media [13] or laminar strips [14], but then an external vacuum-17 production system is needed to switch between the compliant and rigid states which is too 18 complex and bulky to be used for insect-scale robots. Shape-memory materials [15, 16, 17], 19 thermo-rheological fluids as low-melting materials [18, 19] or magnetorheological fluids [20] 20 can also be used to vary the robotic stiffness, but the stimuli required would be environmental 21 heat or magnetic field which also requires bulky system designs. Light-driven systems have 22 also been developed, including using thermally-expanding paraffin wax with the heat input 23 delivered by carbon-nanotubes via a photothermal effect, but in this case the self-stiffening is 24 coupled to the actuation [21]. Some strategies for tuning stiffness in response to electrical 25 signals, which are convenient stimuli for insect-scale robots, have also been proposed [22, 23, 24]. These include the use of shape-memory polymers with the temperature input provided by 26

1 joule heating [23] and electro-active polymers (EAP) [24], but for the former the switching 2 frequencies may be limited by the slow cooling process, and for the latter ultra-high driving 3 potentials in kilovolts would be needed. Furthermore, most of the above strategies are only for 4 controlling the structural stiffness function without motions delivered [15, 16, 17, 18, 19, 20, 23, 24]. For small-scale robotic devices, strategies for stiffness tuning as well as actuation in 5 6 response to electrical signals have also been proposed, including the use of oxygen-group 7 functionalized graphene that exhibits a modulus increment during its tension actuation via 8 thermal expansion by joule heating [25], and intracellular pressure tuning by reversible osmosis 9 triggered by potential-driven electrolytic changes [26], in a synthetic system that bio-mimics 10 the movement of plants [27]. However, in both cases, the stiffening is intrinsically coupled to 11 the actuation straining rather than being independently controllable, and in the case of 12 reversible osmosis, the response is very slow in minutes [26].

13 Here, we introduce a simple, 3D printing strategy to rapidly switch the stiffness of thin-14 film robots on-demand. Inspired by the biomimetic design of insect wings [28], a concept of 15 "curvature-induced rigidity" is proposed (Figure 4a) by imposing a transversal curvature to 16 stiffen a thin sheet against an external bending force [29, 30]. With a versatile open-17 electrodeposition printing method for the oxide-based actuating material, the actuators can be 18 quickly fabricated in typically less than 15 minutes, and in operation they can deliver fast and 19 large bending over 500° within 5 s under an electrochemical potential of <1 V. Beside simple 20 bending, foldable actuators, an independently controlled gripper and a micro-robotic arm with 21 rapidly tunable stiffness are demonstrated. Compared to 4D printed actuators based on 22 hydrogels or shape memory polymers [8], our actuators can produce fast, large, flexible and 23 reversible shape transformation with much quicker stiffness tuning, in a simple structure 24 fabricated by a simple one-step method. Compared to other systems in which the stiffening is 25 intrinsically coupled with the actuation [21, 25, 26], the stiffening in our method is decoupled from the actuation, and the two are independently controllable. 26

1 **2. Experimental Methods**

2 Printed MnO₂ actuating layers. An open-electrodeposition printing setup [31], shown in Figure 3 1a, was used to print the MnO₂ actuating layers. A conductive syringe needle (of 3 ml syringe 4 capacity and 1.27 mm needle external diameter) controlled by an x-y-z motion table was used 5 to deliver a small droplet (~ 0.05 ml) of the electrodeposition electrolyte (0.1 M C₄H₆MnO₄·4H₂O and 0.1 M Na₂SO₄ [12]) at a small gap (~ 2 mm) maintained between the 6 7 needle tip and a metallized substrate, and then the active material of MnO₂ was 8 electrochemically formed within the gap via the application of a constant potential difference 9 (1.8 V to 2.5 V with electrochemical workstation: CS350, Corrtest) between the syringe needle 10 which acted as the reference electrode (RE) and the counter electrode (CE), and the substrate which acted as the working electrode (WE) [31]. In this work, the substrates used were a type 11 12 of 10 µm thick micro-porous PC films (Nuclepore Track-Etched Polycarbonate Membranes, 13 Whatman[®]) metallized by Au sputtering (Bal-tec SCD 005 Sputter Coater) beforehand. The 14 micro-pores on the PC substrates allowed MnO₂ to be grown into them during 15 electrodeposition, thus forming "micro-rivets" to enhance the adhesion between the MnO₂ layer 16 and the substrate [12], as shown in Figure 1c. The printing was performed by moving the 17 syringe needle at a speed of 1 to 500 mm/min. To achieve thicker active layers of MnO₂, the 18 writing was conducted by repeatedly scanning over the same area, for which the average 19 printing time per unit net printed length was measured. For the actuating hinges, the average 20 printing time was 2.4 min/mm, and for the stiffness-enhancing hinges, the average printing time 21 was 0.6 min/mm. (see Figure 4b). For the independent controlled gripper, muscle groups of the 22 MnO₂ material were printed onto isolated Au-sputtered PC areas, which were then individually 23 connected to a multi-control switch and triggered in 1 M Na₂SO₄ solution. Printed MnO₂ was 24 characterized by scanning electron microscopy (SEM) in a Hitachi S-4800 microscope, and 25 transmission electron microscopy (TEM) in an FEI Tecnai G20 Scanning TEM.

26 3. Results and discussion

3.1. Characterization and printed actuating behavior

2 Scanning electron microscopy (SEM) and transmission electron microscopy (TEM) 3 examination (Figure 1b) show that the printed MnO₂ was amorphous and turbostratic [11] with 4 a relatively crumpled microstructure. As mentioned above, MnO₂ was grown into the micro-5 pores of the PC film to form micro-rivets to enhance the adhesion between the MnO₂ actuating layer and the PC substrate film (Figure 1c) [12]. The electrochemical actuation in 1 M Na₂SO₄ 6 7 electrolyte, as shown in Figure 1d, was due to a volume expansion accompanying Na⁺ insertion 8 to form MnOONa in the reduction process, and contraction accompanying recovery to MnO₂ 9 under oxidation, during a reversible redox reaction (Figure 1d) [32]. Figures 1e-g show the 10 effects of the applied constant potential, print-head speed, and the average printing time per line 11 length on repeated scanning, on the actuation performance of the printed actuator. Figure 1e 12 shows that the MnO₂ can only be printed effectively onto the conductive substrate with a 13 printing potential of above ~1.9 V, indicative of a minimum potential threshold for the electrodeposition to happen. An constant potential of around 2.1 ~ 2.3 V applied resulted in 14 15 relatively good actuating performance; although a larger potential may result in a thicker 16 actuating layer because of the faster deposition in a certain time, rough and convex structures 17 may also be resulted because of the faster deposition at the center of the locally printed feature 18 [33], which may then produce decreased or uneven motions during the subsequent actuation. 19 As shown in Figure 1f, a print-head speed of around 5 ~ 50 mm/min resulted in better actuating 20 performance, and faster or lower speeds may not result in a relatively stable micro-electrolyte 21 condition between the printing needle and the conductive substrate, and/or an even diffusion 22 condition for the electrolyte along the printing path. Under a print-head speed of 5 mm/min, for 23 a 1 cm long and ~ 1 mm wide actuating layer that was reversibly scanned for one cycle, the 24 average printing time was ~ 2/(5mm/min) = 0.4 min/mm along the printing track; from the data in Figure 1g, the average deposition rate was ~1.6 to $2.2 \times 10^4 \,\mu\text{m}^3$ /s which is fast compared to 25 other materials printed by the same open-electrodeposition method [33, 34]. The local actuation 26

curvature can be increased by thickening the actuating layer, which can be achieved by
reversibly scanning over the same printed area during electrodeposition (Figure 1g). However,
when the thickness of actuating layer surpassed 4 µm, the actuating performance decreased,
which may be due to the larger cracks formed in thicker actuating layers [10].



Figure 1. Characterization of the printed MnO₂ active layer. (a) Schematic of printing set-up. (b) SEM surface morphology and TEM of printed MnO₂ layer. (c) Schematic of the "microriveting" strategy for enhancing the adhesion between the printed MnO₂ layer and the PC film. (d) Schematic of actuating mechanism and corresponding cyclic voltammetry of the printed MnO₂ layer in 1 M Na₂SO₄ electrolyte with potential window of 0.0 V to 0.8 V at scan rate of 25 mV/s. (e~g) Dependence of actuating curvature of MnO₂ layers on printing parameters including (e) constant potential applied (at print-head speed of 5 mm/min for three cycle of forth and back scanning), (f) print-head speed (at step potential of 2.1 V for three cycle of forth and back scanning), and (g) average printing time per printed line length on repeated scanning for different cycles (at step potential of 2.1 V and print-head speed of 5 mm/min).

1 **3.2.** Foldable, sizable and independently controlled motions

2 Using the local printing method described above, both sides of the substrate can be printed, 3 as shown in Figure 2a. Compared to one-side printing (Figure 2b), the two-side printing offers 4 more applications in origami structures, in which mountain and valley self-folding hinges can 5 deform the substrate in opposite directions in response to a simultaneous stimulus. The open-6 electrochemical printing method also breaks the limit of the traditional bath-electrodeposition 7 method, in which large substrates can only be printed upon in a larger cell, which would then 8 lead to uneven electrodeposition. Figure 2c shows a printed actuator over 5 cm long that 9 exhibited a high actuating performance of curling into multiple loops, demonstrating that rather 10 large and highly functional actuators can be printed by the present open-electrochemical 11 printing method (supporting Movie S1). A gripper was further designed and printed for carrying 12 out the typical robotic task of gripping (Figure 2d). Here, each actuator was connected to a 13 multi-control switch, which allowed the rapid switching between different potential windows 14 to tune the actuation speed and deformation. Such flexibility allowed the gripper to grasp a free-15 shape object (Figure 2d and supporting Movie S2).



Figure 2. Complex motions by the printed MnO₂ hinges. (a~b) Foldable actuators enabled by two-side printing (scale bar: 1 cm). (c) An actuator over 5 cm long curling into loops (scale bar: 1 cm, see supporting video S1). (d) Functional gripper grasping a free-shape object (see supporting video S2).

3.3. Film-actuator with rapidly tunable stiffness

2 The thin passive layer would enable a large local curvature [12], but as shown in Figure 3, 3 the intrinsic softness of a thin film would make the motion of the actuator hard to control 4 effectively and lower the load-bearing capability. As shown in Figure 3a, large actuation at the 5 elbow can be delivered by the printed actuating hinges, but the arm deforms significantly due 6 to its gravitional load during the actuation process. Finite element method (FEM) (ABAQUS, 7 Dassault Systèmes) simulation in Figure 3b also shows significant bending of the over-hanging 8 passive layer (supposed to be the robotic arm) due to its own weight, despite the large actuation 9 produced by the actuating hinges at the elbow.



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Figure 3. Compliant micro-robotic arm without curvature-induced rigidity. (a) Experimental and (b) FEM results.

11 Contrary to the above, dragonfly's wing is very thin (< 10 µm) but is capable of 12 withstanding large loads by vurtue of a corrugated structure that produces stiffening [28]. This 13 effect is also found in the beetle's hind wing [35], where the camber of the wing could stiffen 14 itself. Figure 4a shows the stiffness-enhancement concept in such cases: for an elastic strip, 15 formation of a curvature (k) about the long axis at the free end will enable the strip to endure 16 its gravitional load, and increasing the curvature can rigidify the strip against the weight (g)17 better (Figure 4a) [29, 36]. Inspired by the intelligence of insect wings, in our present thin-film 18 based actuators, the curvature k_v for increasing rigidity was produced by another set of MnO₂ active hinges printed along the long axis of the film as shown in Figure 4b, in addition to the 19 20 main MnO₂ actuating hinges that were printed in the transverse direction for the bending actuation of the device. For a thin-film strip with original bending stiffness $S_0 = 4Ebh^3/L^3$ 21

1 where E is Young's modulus, b is width, h is thickness and L is length, the change in stiffness

2 (ΔS) due to transversal and longitudinal curvatures, k_y and k_x respectively, is given by [29, 30]

$$\frac{\Delta S}{S_0} = \frac{b^4}{60h^2} \left(k_y - \nu k_x \right)^2$$
(1)

3 where v is the Poisson ratio. When the other end of the strip is clamped as shown in Figure 4a, 4 an extra coefficient of $e^{-c\frac{b}{L}}$ is involved:

$$\frac{\Delta S}{S_0} = \frac{b^4}{60h^2} \left(k_y - \nu k_x \right)^2 e^{-c\frac{b}{L}}$$
(2)

5 where $c \approx 3.078$ [29].

6 Also, Figure 1g indicates that increasing the thickness of the actuating layer would in 7 general result in a larger local curvature. The intrinsic actuating stress σ_a and strain ε_a of 8 actuators made with different MnO₂ thickness were calculated using the Timoshenko strain-9 mismatch theory for bi-layers as shown in Figure 5a [12, 37]. It can be seen that the intrinsic 10 actuating stress σ_a of MnO₂ is fairly constant at about 13±3 MPa in the range studied, but the 11 actuating strain ε_a increases with the average printing time and hence the thickness of the 12 actuating layer, which is understandable as the substrate thickness is a constant factor so a 13 thicker active layer should actuate more. The experimentally derived data in Figure 5a was used 14 as input to simulate the curvature-induced-rigidity effect by the FEM, so as to guide our 15 experimental design of the robotic arm in Figure 4b. The simulation setup is shown in Figure 16 5b. The MnO₂ and PC film were modeled as purely elastic solids with Young's modulus of 35.3 17 GPa [32] and 1.5 GPa (by tensile test), respectively. The electrochemical actuation was 18 modelled as a thermal contraction problem in the FEM - to reproduce the experimentally 19 measured actuating strains in Figure 5a, the thicker actuating hinges, thinner stiffness-20 enhancing hinges, and the PC substrate were modelled as thermally contracting solids with 21 linear thermal expansion coefficients of ~1 %, ~0.3% and zero, respectively, under a negative 22 temperature change of 1 degree imposed. Perfect bonding was assumed between the printed 23 MnO₂ actuating hinges and the PC membrane. Since the static equilibrium response instead of

1 dynamic behaviour was calculated, fluidic drag effects of the electrolyte were not considered, 2 and buoyance was also neglected. As shown in Figure 3b, in the absence of stiffness-enhancing 3 actuating hinges, the simulated transverse curvature k_y of the micro-robotic arm is zero as 4 expected. With a group of stiffness-enhancing hinges (intrinsic actuating strain ~0.34%, thickness ~0.6 μ m) added as shown in Figures 5b and 4c, the simulated curvature k_v is ~ 0.28 5 mm⁻¹. According to the simulation result, for the experimental design of a micro-robotic arm in 6 7 Figure 4d, the stiffness-enhancing pattern was printed with an average printing time of ~ 0.6 8 min/mm (intrinsic actuating strain ~0.34%, thickness 0.6 µm) to deliver an approprite transverse 9 curvature, and the thicker actuating hinges were printed with an average printing time of ~ 2.4 10 min/mm (intrinsic actuating strain ~1.2%, thickness 3.25 µm) to deliver large local deformation 11 at the arm's elbow for load bearing. The experimental transverse curvature was measured to be 12 about 0.26 mm⁻¹, in good aggreement with the simulation result.

13 Pieces of solid plasticine (density $\sim 1.9 \text{ g/cm}^3$) of different weights were applied to test the 14 load-bearing capablity of the whole micro-robotic arm. In this procedure, the actuating unit was 15 first switched on to lift up the robotic arm, and then the plasticine was added bit by bit onto the 16 free end of the robotic arm until the arm became low enough to let the plasticine load drop off from it. The maximum loads thus recorded were 10.3 µN and 40.5 µN after accouting for the 17 18 buoyancy, without and with the stiffness-enhancement unit, respectively. The load-bearing 19 capability of the robotic arm was also calculated by FEM – as shown in Figure 5b, a node at 20 the tip of the actuator is constrained to zero displacement, and as the actuating hinges actuate, the force at the constrained node was calculated to be 11.5 µN and 35.7 µN, respectively, 21 22 without and with the stiffness-enhancing hinges turned on. The FEM predictions therefore agree 23 with the experimental results (see table in Figure 4e) within around 10%, which may be due to 24 the slightly uneven deposition during the printing process.

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Figure 4. Curvature-induced rigidity in a micro-robotic arm. (a) Concept of curvatureinduced rigidity, with the stiffness of a strip enhanced by a transverse curvature k at the free end to increase the bearing capability of weight (g) [29, 36]. (b) Schematic of the strategy for the printed actuator design comprising thick actuating hinges and thin stiffness-enhancing hinges, showing bending angles θ_1 and θ_2 , and transverse and longitudinal curvatures k_y and k_x . (c) Finite element model and (d) experimental results of the rigidity-enhanced microrobotic arm, with the intrinsic actuating strains for thinner and thicker actuating hinges being 0.34% and 1.25%. (e) Experimental, FEM and theoretical results for the load-bearing capability of the micro-robotic arm without and with stiffness-enhancement units. (f) Fastresponsive micro-robotic arm with real-time bending angles θ_1 and θ_2 (supporting Movie S3) and (g) good stability over ~ 400 cycles (scale bar: 1 cm).



Figure 5. FEM simulation of load-bearing capability. (a) The intrinsic actuating stress and strain for the printed MnO₂ layers prepared from different average printing times. Actuating curvature is calculated from the actuating stress stress σ_a is calculated from the actuating curvature k_x via $\sigma_a = k_x/[6h_a(h_a + h_s)/(E_ah_a^3 + E_sh_s^3)]$, in which $h_s = 10 \ \mu\text{m}$, $E_s = 1.5 \ \text{GPa}$ (by microtensile σ_a k_r via $\sigma_a =$ test), $E_a = 35.3$ GPa [32], and actuating strain is calculated from $\varepsilon_a = \left[\frac{1}{E_a} + \frac{h_a}{E_s h_s} + \frac{h_a}{E_s h_s}\right]$ $\frac{3h_a(h_a+h_s)^2}{E_ah_a^3+E_sh_s^3}$] σ_a [12]. (b) Geometry design of FEM for the actuating arm with stiffnessenhancing hinges. For the PC film, the length, width and thickness are 28 mm, 8 mm and 0.01 mm, respectively. For the actuating hinges, the length, width and thickness are 8 mm, 1.2 mm and 0.003 mm, and for the stifness-enhancement hinges, the length, width and thickness are 6~8 mm, 1.2 mm and 0.0006 mm. For calculating the load-bearing capacity, a node (marked red) is constrained to have zero displacement along the constrained direction, and as the actuating hinges actuate, the calculated forces at the constrained node along the constrained direction with and without the stiffness-enhancement units are 35.7 µN and 10.7 μN, respectively.

1 For the micro-robotic arm without the stiffness-enhancing unit, the load-bearing capability 2 is limited by the intrinsic softness of the PC substrate. Here, adding the maximum sustainable 3 weight of 10.3 μ N was found to cause the substrate to deflect by $\Delta x \sim 5$ mm (Figure 6a). The intrinsic bending stiffness of the PC substrate calculated by $S_0 = 4Ebh^3/L^3$ is ~2.2 µN/mm, and 4 5 for the deflection Δx of ~5 mm, the load is estimated to be ~5 × 2.2 = 11.0 µN (Figure 4e), in 6 good agreement with the actual value of 10.3 µN. Experimentally, when the stiffness-enhancing 7 unit was turned on, the arm portion of the robot became very rigid, as can be seen by the 8 demonstration in Figure 6b, where a much-heavier plastine of dry weight ~ 700 mg before 9 accouting for bouyance was used to push the micro-robotic arm downward. Here, after 10 switching on the stiffness enhancing unit, the arm portion did not exhibit any noticeable 11 deformation under the external load. Also, according to Equation (2), as the transversal 12 curvature k_v is imposed the stiffness due to curvature-induced rigidity is increased to ~8.8 N/m.

1 At such a stiffness, an end-deflection of $\Delta x = 5$ mm as in Figure 6a would be produced by a load of ~ 44 mN, which is ~ 4000 times of the case without the stiffness-enhancers. Therefore, 2 3 with the stiffness-enhancing unit in action, the load-bearing capability is limited by the intrinsic 4 load-lifting capability of the actuating unit itself, since the arm portion would be close to rigid. Under no external load, the actuating unit was observed to produce a curvature k_x of 0.34mm⁻¹ 5 (Figure 6b). The bending moment of the actuating unit per unit width is given by M =6 $\frac{(E_a h_a^3 + E_s h_s^3)k_x}{12}$, where E_a and E_s are the Young's moduli of the active material and the substrate, 7 and h_a and h_s are their thicknesses [3]. The maximum load capacity P_a of the robotic arm can 8 9 be thought of as the load at the free end needed to straighten the actuating unit against its elasticity back to zero curvature, and hence $P_a \times L' = M \times b$ where the lever arm $L' \approx 15$ mm 10 11 (see Figure 4e). The load-bearing capacity of the robotic arm, with the stiffness-enhancers in action, is then calculated to be $P_a = 40.0 \,\mu\text{N}$. As shown in Figure 4e, the theoretical load-bearing 12 13 capacities with and without the stiffness enhancers are in good agreement with the experimental 14 and FEM results.

15 As shown in Figure 4f, the micro-robotic arm with curvature-induced rigidity can be 16 rapidly transformed in response to ultra-low voltage applied (~ 1V) within 5 seconds. The 17 measured transverse folding angle θ_1 is larger at the free end than near the folding hinges, and 18 as the bending energy is related to the transverse curvature, the small transverse curvature near 19 the folding hinges is beneficial to the effective bending during the actuating motion. Finally, 20 cyclic actuation tests were performed to evaluate the reliability of the printed micro-robotic arm, 21 and as shown in Figure 4g, the robotic arm demonstrates stable actuation over at least 400 cycles 22 without obvious degradation.

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¹ Figure 6. Comparison of robotic arm with and without stiffness-enhancing units. (a) Deflection Δx of the micro-robotic arm without the stiffness-enhancing units due to a load of 10.3 µN after accouting for buoyancy. (b) With the stiffness-enhancing units, the micro-robotic arm does not exhibit significant self-deformation under a load of ~330 mg after accouting for buoyancy.

2 **4. Discussion**

3 In this work, the critical problem of how to stiffen a thin-film robotic device in a fast, 4 compact and easily controllable way was solved by a minimalist design concept of curvature-5 induced rigidity, in which a high-performing stimulus-responsive material is directly printed on 6 pre-designed locations of the robot body to induce local curvatures to stiffen it. In particular, 7 the high-performing, electrochemical actuating material of MnO₂ was employed as both the 8 actuating muscles and the stiffness enhancers, leading to large actuation response with much 9 improved load-bearing capability of the device, without sacrificing dexterity. Compared to the 10 other stiffening strategies involving electrical stimuli (see Table 1), our actuating-stiffening 11 method has the advantages of low electrical input (~ 1 V), fast response (~5 s), as well as 12 uncoupled actuation and stiffening based on a single functional material driven by one type of 13 stimuli. As for fabrication, existing methods for making metallic patterns include the selective-14 laser sintering [38], the ink-jet printing [39] or selective electrodeposition methods [40, 41]. However, sintering is not suitable for the present type of active materials which are high-15

1 melting oxides (MnO₂) or the substrate materials which are low-melting polymeric films, and 2 the extra mask or the sacrificial layer involved in masked deposition methods may increase the 3 cost and complexity of the fabrication process. On the other hand, the present printing 4 fabrication method offers a great deal of flexibility in terms of the deposited materials, pattern 5 geometry and thickness, thus allowing muscle and stiffening groups to be individually printed 6 for independent electrical control, so that compact insect-scale robotic devices demonstrating 7 complex motions and tunable stiffness can be easily fabricated. This printing method is 8 applicable not only to the present MnO₂ material but also to other electrochemically printable 9 actuating materials [2, 3, 9]. The motion of the resultant actuating device can be controlled by 10 modifying the scan rate as well as the potential window applied during the printing of the 11 actuating material. However, for effective electrochemical actuation of the present active 12 material, the device must be submerged into the electrolyte instead of floating on it. Potential 13 applications of the present actuating-stiffening concept may include marine (because of the role 14 of Na⁺ions) biomimetic devices such as octopus-mimicking robots which require easy morphic 15 changes of their bodies as well as the formation of temporary skeletons by self-stiffening [22]. 16 Table 1 – Stiffness-tuning methods for insect-scale robots driven by electrical signals

Stiffening system	Input requirem ent	Response speed	Coupled with actuation	Other remarks
Shape memory polymers [17, 23]	1.4 A [<i>17</i>]; 20~50 V [2 <i>3</i>];	Transition from solid to soft state is quick (in seconds), but reverse transition by cooling is slow	No. Actuation needs to separately provided (e.g., by a pneumatic system)	Complicated assembly process
Electroactive polymers	kV range	Quick, less than 1 s	No.	-
[24]			Tunable stiffness for structure only	
Graphene functionalized with oxygen-rich groups [25]	~5 V	10 s	Coupled with actuation without independent control	Complicated preparation process
Reversible osmosis [26]	~1 V	Slow in minutes	Coupled with actuation without independent control	Biomimicking

Curvature-induced < 1 V in ~ 5 sec rigidity using printed stimuli-responsive materials (present work) Based on same principle as actuation, but stiffening and actuation are separately controllable Applicable to all types of stimuliresponsive materials that are printable

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2 **5.** Conclusion

3 In summary, a versatile yet simple paradigm for fabricating micro-robots with fast 4 response and independently controllable motions was established. By a low-cost desktop open-5 electrodeposition method, rapidly fabricated actuators capable of producing large, fast and 6 complex motions in response to ultra-low electrochemical stimuli were demonstrated. 7 Moreover, leveraging on the flexibility of the printing method, the actuating hinges not only 8 rendered a motion-delivering function, but also a novel, on-demand structural stiffening 9 function with a theoretical improvement of stiffness by ~4000 times. This study provides a 10 simple technology for achieving self-stiffening as well as self-actuating functions for insect-11 scale thin-film structures.

12 Supporting Movies

13 Movie S1. A large-size curling actuator.

14 Movie S2. Functinoal gripper capable of grasping and manipulating the free-shape object.

15 Movie S3. The arm actuator with curvature-induced rigidity.

Author contributions: A.H.W.N. supervised the overall research. K.W.K. and R.W. conceptualized the printed MnO₂ actuator and curvature-induced actuating system. R.W. performed experiments and collected the data. R.W., K.W.K. and A.H.W.N. did the analysis and wrote the paper. Funding: The work is funded by the Kingboard Endowed Professorship in Materials Engineering at the University of Hong Kong. Acknowledgements: We thank Mr F.Y.F. Chan for help with TEM sample preparation and TEM operation. Competing interests: The authors declare that they have no competing interests.

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