

1 **Hydrophilic Silver Nanoparticles Induce Selective**
2 **Nanochannels in Thin Film Nanocomposite Polyamide**
3 **Membranes**

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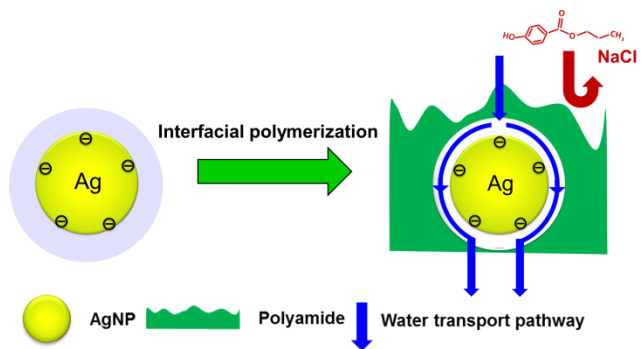
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16 **ABSTRACT**

17 Thin-film nanocomposite (TFN) membranes have been widely studied over the past
18 decade for their desalination applications. For some cases, the incorporation of
19 nonporous hydrophilic nanofillers has been reported to greatly enhance membrane
20 separation performance, yet the underlying mechanism is poorly understood. The
21 current study systematically investigates TFN membranes incorporated with silver
22 nanoparticles (AgNPs). For the first time, we revealed the formation of nanochannels
23 of approximately 2.5 nm in size around the AgNPs, which can be attributed to the
24 hydrolysis of trimesoyl chloride monomers and thus the termination of interfacial
25 polymerization by the water layer around each hydrophilic nanoparticle. These
26 nanochannels nearly tripled the membrane water permeability for the optimal
27 membrane. In addition, this membrane showed increased rejection against NaCl,
28 boron, and a set of small-molecular organic compounds (e.g., propylparaben,
29 norfloxacin, and ofloxacin), thanks to its combined effects of improved size exclusion,
30 enhanced Donnan exclusion, and suppressed hydrophobic interaction. Our work
31 provides fundamental insights into the formation and transport mechanisms involved
32 in a solid-filler incorporated TFN membranes. Future studies should take advantage of
33 this spontaneous nanochannel formation in the design of TFN for overcoming the
34 classical membrane permeability-selectivity tradeoff.

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40 INTRODUCTION

41 Thin-film composite (TFC) polyamide membrane, formed by the interfacial
42 polymerization (IP) reaction between *m*-phenylenediamine (MPD) and trimesoyl
43 chloride (TMC),¹ is the core to reverse osmosis (RO) technology applied to seawater
44 desalination wastewater reclamation.^{2, 3} Its separation performance has a profound
45 impact on the RO process.¹ Specifically, a membrane with high water permeability
46 and selectivity can simultaneously decrease the specific energy consumptions and
47 increase the permeance quality.⁴ Unfortunately, the separation performance of
48 commercial TFC RO membranes are constrained by the upper bound, a well-known
49 tradeoff between membrane water permeability and selectivity.^{5, 6} Thin-film
50 nanocomposite (TFN) membrane that incorporates nanomaterials into the polyamide
51 rejection layer is an effective strategy to overcome this permeability-selectivity
52 tradeoff. In 2007, Hoek and co-workers⁷ first reported the use of porous zeolite
53 nanoparticles (NPs) to provide additional pathways to water in a polyamide rejection
54 layer without compromising its salt rejection. After that, many other porous
55 nanomaterials, such as mesoporous silica^{8, 9}, MOFs^{10, 11}, and aquaporins^{12, 13} have
56 been explored to enhance membrane separation performance.

57

58 In the last decade, researchers have also investigated the incorporation of solid
59 nanofillers, such as silica,^{14, 15} titanium dioxide,^{16, 17} zinc oxide,^{18, 19} cerium oxide,²⁰,
60 carbon quantum dots (CQDs),^{21, 22} and silver nanoparticles (AgNPs)²³⁻²⁵ to make TFN

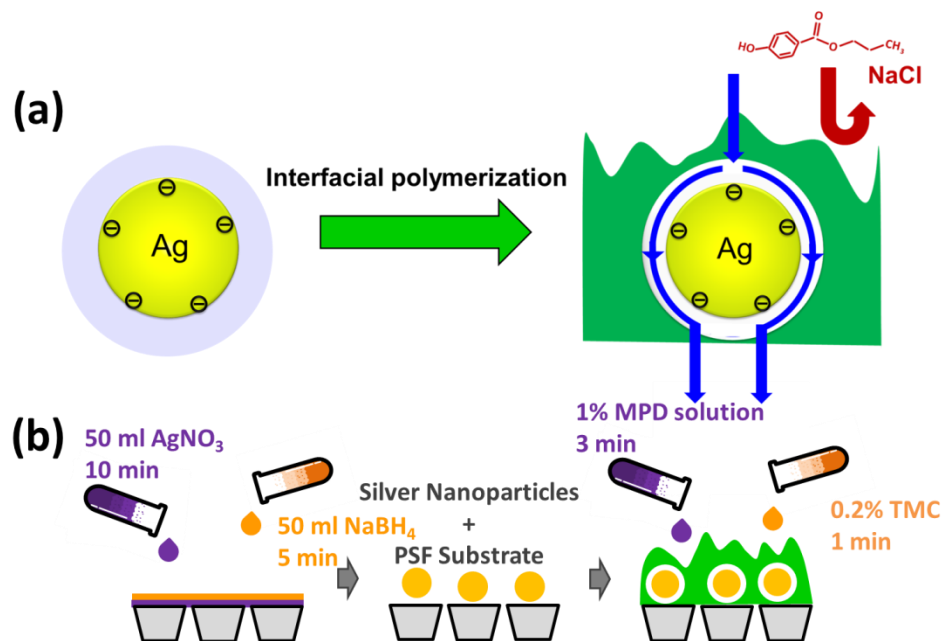
61 membranes. Interestingly, many studies reported enhanced water permeability of their
62 TFN membranes,¹⁴⁻¹⁸ even though the nanomaterials contain no internal
63 pores/channels. Researchers often attribute this flux enhancement to the
64 hydrophilicity of the embedded nanofillers.²⁶ However, little is known about the
65 nanoscale transport mechanism that leads to such enhancement. It is also worthwhile
66 to note that several studies reported an opposite trend: decreased water fluxes upon
67 the incorporation of solid nanofillers in TFN membranes.^{12, 25, 27, 28} In addition to
68 water flux, contradictory results related to salt rejection of solid nanofillers
69 incorporated TFN membranes, either increased^{23, 29} or declined,^{27, 30, 31} have been
70 reported (Table 1). Despite the dozens of papers published on this topic,³² the
71 apparently disparate phenomenological results prompt us to perform a systematic
72 investigation on the underlying mechanism governing the transport in solid nanofillers
73 incorporated TFN membranes.

74

75 In this study, we performed detailed characterization of TFN membranes
76 incorporating AgNPs as nanofillers. Presumably, the highly hydrophilic nature of
77 AgNPs can lead to the formation of a hydration layer around each nanoparticle (³³ and
78 Figure 1a). With water molecules acting as terminators by hydrolyzing TMC, the
79 presence of this water layer may prevent the formation of polyamide in the vicinity of
80 the fillers. Therefore, we hypothesize that the addition of hydrophilic nanofillers into
81 the aqueous MPD solution during interfacial polymerization induces the formation of

82 nanosized channels that facilitate water transport. For the first time, we provide
83 compelling and direct microscopic evidence of the presence of nanochannels in an
84 AgNPs-loaded TFN membrane. The role of these nanochannels on the transport of
85 water, salt and micropollutants was systematically elucidated. Our study provides
86 deep mechanistic insights for guiding the selection of nanofillers and design of TFN
87 membranes.

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92 Figure 1. (a) Schematic diagram of the mechanism of AgNPs induced nanochannels in the
93 polyamide layer for efficient water transport, (b) schematic diagram of membrane fabrication. A
94 polysulfone substrate was soaked by 50 ml AgNO₃ solution and 50 ml NaBH₄ solution
95 respectively to generate the AgNPs. Then, the interfacial polymerization reaction between 1%
96 MPD and 0.2% TMC was performed on this AgNPs loaded substrate to obtain the final AgNPs
97 incorporated TFN membranes.

98

99 MATERIALS AND METHODS

100 **Materials and Chemicals.** Unless specified otherwise, all reagents and chemicals
101 were analytical grade and were purchased from Sigma-Aldrich. Polysulfone (PSF,
102 Molecular weight ~35,000) and *N,N*-dimethylformamide (DMF, 99.8%, anhydrous)
103 were used for preparing substrates. Silver nitrate (AgNO_3 , ACS reagent, $\geq 99.0\%$) and
104 sodium borohydride (NaBH_4 , 98%) were applied to generate AgNPs. MPD (flakes,
105 99%), TMC(99%, J&K Scientific Ltd.) and hexane (HPLC grade, 95%) were used for
106 synthesizing the polyamide rejection layer through interfacial polymerization reaction.
107 Sodium chloride (NaCl) was purchased from Uni-Chem and boron, ethylene glycol,
108 diethylene glycol, and pentaerythritol were obtained from Dieckmann Hong Kong.
109 Propylparaben (99%, Mw 180.2, Acros Organics, Geel, Belgium), norfloxacin (NOR,
110 Mw 319.3) and ofloxacin (OFL, Mw 361.4) were used as model trace organic
111 contaminants (TrOCs). Propylparaben is an endocrine disrupting compounds, whereas
112 both NOR and OFL are antibiotics.

113

114 **Synthesis of Control PSF and AgNPs Coated PSF Substrates.** Polysulfone
115 (PSF) substrate was fabricated using a conventional phase inversion method following
116 our previous work.³⁴ Due to their highly hydrophilic nature and ability to be easily
117 generated *in situ* with minimized particle aggregation, AgNPs were used as model
118 hydrophilic nanofillers in the current study.³⁵ AgNPs were generated *in situ* on the
119 polysulfone substrate (Figure 1b) following the method reported in the literature.³⁵

120 Briefly, a 50 ml AgNO₃ solution (1, 5, 20 or 100 mM) was poured onto a polysulfone
121 substrate, and shaking was applied for 10 mins at 50 revolutions per minute (rpm).
122 The extra AgNO₃ solution was removed. Then, a 50 ml NaBH₄ solution (with
123 identical concentration to the AgNO₃ solution) was applied to the AgNO₃ soaked
124 polysulfone substrate to reduce Ag⁺ to AgNPs for 5 mins. The AgNPs-modified
125 substrate was subsequently rinsed by deionized (DI) water for 5 mins and stored in DI
126 water at 4 °C before use. The PSF substrates prepared in this study were named as
127 PSF-Ag1, PSF-Ag5, PSF-Ag20 and PSF-Ag100 on the basis of the AgNO₃
128 concentration.

129

130 **Synthesis of the TFC and TFC-Ag polyamide membranes.** To obtain the TFC
131 membrane, interfacial polymerization process was applied onto the control
132 polysulfone and PSF-Ag substrates. Specifically, an AgNPs modified substrate was
133 soaked in a 100 ml MPD/deionized (DI) water solution (1 wt%). The immersion time
134 was set at 3 mins and extra MPD solution was removed. Then, a 25 ml TMC/hexane
135 solution (0.2 wt%) was poured onto the polysulfone substrate with MPD to perform
136 the interfacial polymerization reaction. The reaction time was set at 1 min in order to
137 obtain the polyamide layer. The resulting membrane was washed by hexane and
138 baked at 60 °C and then stored in DI water at 4 °C before use. The membranes were
139 denoted as TFC-Ag1, TFC-Ag5, TFC-Ag20, and TFC-Ag100 based on the AgNPs
140 loading condition.

141

142 **Membrane Characterization.** The surface morphologies of the polysulfone
143 substrates and TFC membranes were assessed by scanning electron microscopy (SEM,
144 LEO1530, FEG UK) at 5 kV. Before characterization, samples were dried and coated
145 by platinum and gold. Transmission electron microscopy (TEM, Philips CM100) was
146 used to examine the cross-sectional images of the polyamide membranes according to
147 our previous study.³⁶ Briefly, membrane samples were embedded in a resin (Epon,
148 Ted Pella, CA). After curing, the resin block was sectioned by an Ultracut E
149 ultramicrotome (Reichert, Inc. Depew, NY) into TEM sections of 100 nm in thickness.
150 These sections were picked up onto a copper grid and further examined in TEM at an
151 accelerated voltage of 100 KV. Size distributions of the observed nanochannels were
152 obtained using image analyzing software (Image-Pro Plus 6.0, MediaCybernetics, Inc).
153 The detailed information of attenuated total reflection Fourier transform infrared
154 (ATR-FTIR) for analyzing membrane surface functional groups, atomic force
155 microscopy (AFM) for membrane roughness evaluation, X-ray photoelectron
156 spectroscopy (XPS) for membrane surface elemental composition quantification,
157 streaming potential for membrane zeta potential measurement and quartz crystal
158 microbalance with dissipation (QCM-D) for the sorption test can be found in our
159 previous work.^{23, 27, 37}

160

161 **Separation Performance Testing.** Membrane water flux and solutes rejection
162 were evaluated by a cross-flow filtration setup based on our previous method.²⁷ The

163 membrane was filtrated by DI water at an applied pressure of 21 bar for at least 3 h to
164 achieve a stable water flux. When measuring the water flux, the pressure was reduced
165 to 20 bar, and water flux was calculated based on Equation (1).

$$166 \quad J_w = \frac{V}{A \times \Delta t} \quad (1)$$

167 where J_w ($\text{Lm}^{-2}\text{h}^{-1}$) is the pure water flux, V (L) is the volume of permeate, A (m^2) is
168 the membrane area and Δt (h) is the testing time. Then, NaCl was added to the feed
169 solution at 2000 ppm. The filtration test was continued for another 3 h and NaCl
170 rejection (Equation (2)) was determined by measuring the conductivity of the feed and
171 permeate solution with a conductivity meter (Ultrameter II TM, Myron Company,
172 Carlsbad, CA).

$$173 \quad R = \left(1 - \frac{C_p}{C_f}\right) \times 100\% \quad (2)$$

174 where R is the salt rejection, where C_p and C_f are the conductivity of the permeate and
175 original feed solution, respectively.

176

177 The water permeability was based on Equation (3), while the salt permeability was
178 calculated using Equation (4).²⁷

179

$$180 \quad A = \frac{J_w}{\Delta P - \Delta \pi} \quad (3)$$

181
$$B = \frac{1-R}{R} J_w$$
 (4)

182

183 where A ($\text{Lm}^{-2}\text{h}^{-1}\text{bar}^{-1}$) is the membrane water permeability, $\Delta\pi$ is the difference of
184 osmotic pressure between feed and permeate, ΔP is the applied hydraulic pressure,
185 and B ($\text{Lm}^{-2}\text{h}^{-1}$) represents the membrane salt permeability,

186

187 To assess the membrane separation performance, we further measured the rejection of
188 a set of neutral hydrophilic probes (ethylene glycol, diethylene glycol, and
189 pentaerythritol), an inorganic micropollutant (boron), and organic micropollutants
190 (propylparaben, norfloxacin, and ofloxacin). The rejection of ethylene glycol,
191 diethylene glycol, and pentaerythritol (dosed at 200 ppm each in the feed solution)
192 was measured at 20 bar following similar procedures of NaCl rejection measurement.
193 Total organic carbon (TOC, Aurora 1030, OI Analytical, College Station, TX) was
194 used to analyze the concentration of ethylene glycol, diethylene glycol, and
195 pentaerythritol in the feed and permeate samples. The determination of Boron
196 rejection (5 ppm, at pH 6) was similarly conducted, in which an inductive coupled
197 plasma optical emission spectrometer (ICP-OES, Optima 8×00, PerkinElmer) was
198 used to analyze the boron concentration.³⁸

199

200 The rejection of organic micropollutants (propylparaben, norfloxacin, and ofloxacin)

201 was also evaluated. Following the membrane pre-compaction by DI water, a 1 mL
202 stock solution (1 g/L) of each compound was spiked into the 5 L feed solution to form
203 a concentration of 200 µg/L. The filtration process was then continued for another 12
204 h. Subsequently, samples from the feed solution and permeate were collected for
205 further detection of these compounds using an UPLC-MS/MS system.³⁹ The LC
206 separation was implemented on an Agilent 1290 system (Santa Clara, CA) using an C18
207 column (Eclipse Plus, Agilent, 2.1 × 50 mm, 1.8 µm particle size) and the MS/MS
208 quantification was achieved using an API3200 mass detector (AB Sciex, MA).

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211 RESULTS

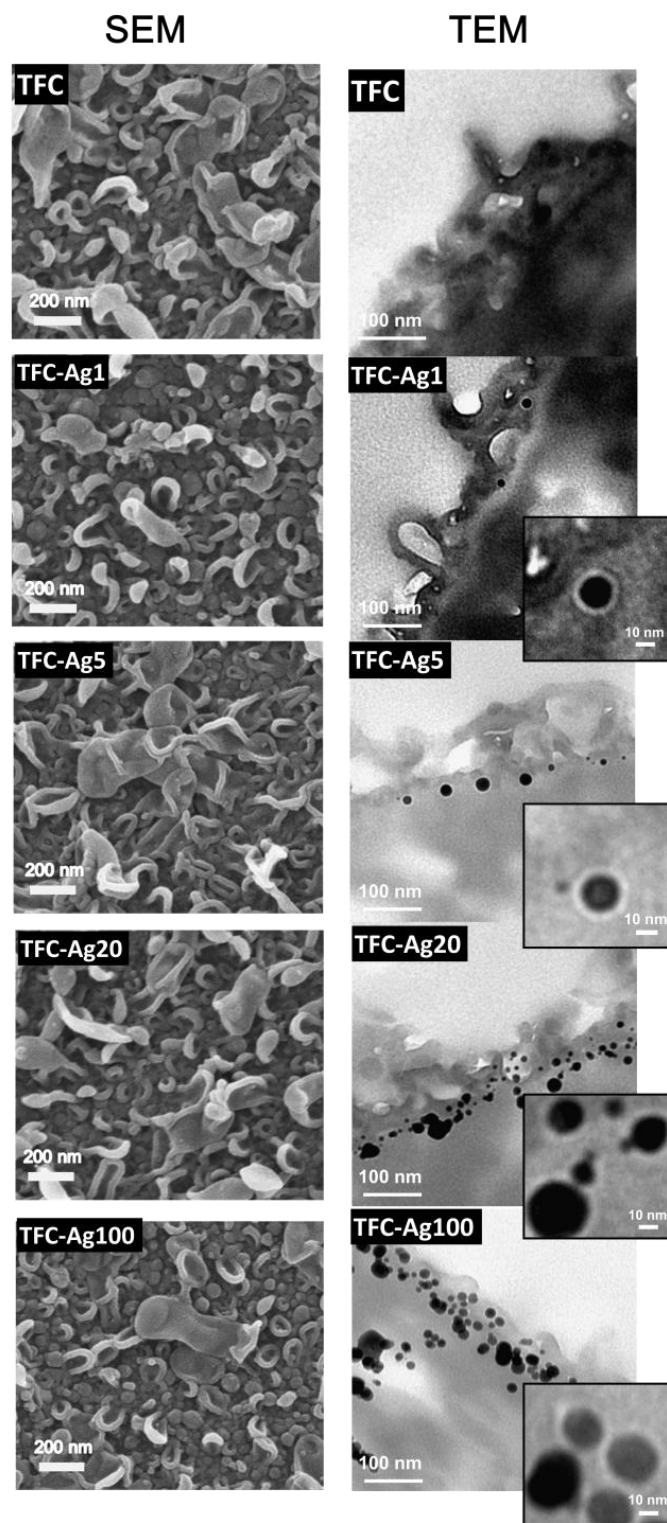
212 **Silver nanoparticles induce nanochannel formation.** SEM micrographs (Figure
213 2, left panel) show that all the polyamide membranes had ridge-and-valley surface
214 morphology that is often observed for RO membranes.^{40, 41} FTIR results (Figure S5 in
215 the Supporting Information) showed the characteristic peaks at 1541 cm^{-1} , 1609 cm^{-1}
216 and 1663 cm^{-1} , representing the Amide II band, the aromatic N-H deformation
217 vibration, and the Amide I band, respectively.⁴² TEM cross-sectional images (Figure 2,
218 right panel) also showed the rough polyamide rejection layer, which are in good
219 agreement with the SEM observations. In addition, the TFC-Ag membranes showed
220 increased presence of AgNPs at higher concentration of the AgNO_3 , which is
221 consistent with the ICP analysis (Figure S1). At relatively low AgNP loading, no
222 severe aggregation of AgNPs occurred (e.g., see the high resolution TEM for
223 TFC-Ag20 in Figure S7a). Nevertheless, aggregation of AgNPs became obvious for
224 TFC-Ag100 (see its TEM cross-section in Figure 2) due to the high AgNO_3 used to
225 generate the AgNPs.

226

227 Interestingly, we observed the presence of a light-colored ring around each AgNPs in
228 the magnified TEM cross-sections of the TFC-Ag membranes (insets of Figure 2, left
229 panel; also see Figure S7b). Such rings were not found in the control TFC membrane.
230 In TEM micrographs, light colored region corresponds to lower mass density.⁴³
231 Therefore, we attribute the rings in the TFC-Ag membranes to the presence of

232 nanochannels around the AgNPs. The TEM characterization directly supports our
233 hypothesis: the presence of hydrophilic nanoparticles induce the formation of
234 nanochannels. In the current study, the hydrophilic AgNPs (Figure S1) could attract
235 water molecules around them, which serves to hydrolyze TMC and terminate the
236 interfacial polymerization in their vicinity (Figure 1a). Further analysis suggests that
237 the size of these nanochannels was approximately 2.5 nm and was nearly independent
238 on the silver loading (Figure S3).

239



240

241 Figure 2. SEM plan view (left panel) and TEM cross-sections (right panel) of the control TFC
 242 and TFC-Ag membranes (magnified micrographs of the AgNPs nanochannels showed in the
 243 insets of the corresponding TEM images).

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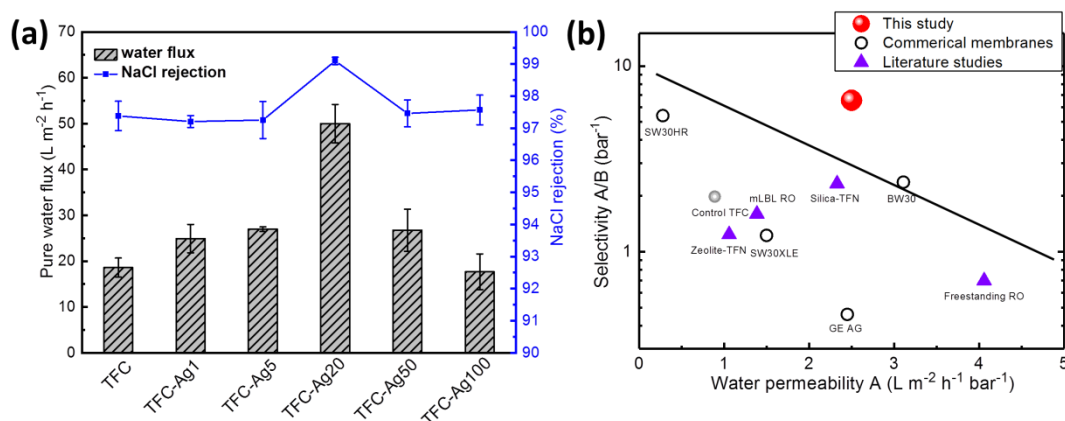
245 **AgNPs-induced nanochannels enhance membrane separation performance.**

246 Figure 3a presents the separation performance of the control and TFC-Ag membranes.
247 With the exception of the TFC-Ag100, increasing the AgNPs loading significantly
248 enhanced membrane water flux. Specifically, the TFC-Ag20 membrane showed the
249 highest water flux of $50.0 \pm 4.2 \text{ L m}^{-2}\text{h}^{-1}$, which was approximately 170% higher than
250 that of the control TFC membrane ($18.6 \pm 2.1 \text{ L m}^{-2}\text{h}^{-1}$, Figure 3a). Such a significant
251 enhancement can be attributed to the AgNPs-induced nanochannels, which act as
252 shortcuts for water transport to significantly reduce membrane hydraulic resistance.²⁷
253 In addition, the hydrophilic nature of AgNPs may also be beneficial to facilitate water
254 transport.³⁶ The TFC-Ag20 membrane also showed the highest NaCl rejection of 99.1
255 $\pm 0.1\%$, compared to that of the control TFC membrane of $97.4 \pm 0.5\%$. In the current
256 study, the size of the nanochannels (approximately 2.5 nm, Figure S3) was
257 significantly larger than the hydrated size of Na^+ (0.358 nm) and Cl^- (0.270 nm).⁴⁴
258 Nevertheless, these nanochannels were narrower than the typical Debye length under
259 the prevailing ionic environment (e.g., approximately 16.5 nm for a permeate water of
260 ~ 20 ppm by assuming a rejection of 99%, Supporting Information S5).^{45, 46} Thus, the
261 AgNPs could play an important role in enhancing NaCl rejection due to the Donnan
262 exclusion effect.⁴⁷ In addition, the improved NaCl rejection of TFC-Ag20 is partially
263 attributed to its improved crosslinking (Figure 4). The presence of AgNPs enhanced
264 the sorption of MPD on the PSF-Ag20 substrate during the interfacial polymerization
265 stage (Figure 4a), possibly due to the synergistic effect of hydrophilic interaction and
266 coordination between MPD and AgNPs.^{48, 49} This greater MPD sorption led to

267 increased crosslinking degree (Figure 4b),⁵⁰ which is beneficial for improving
 268 membrane rejection by enhanced size exclusion. Overloading of AgNPs (e.g.,
 269 TFC-Ag100) led to simultaneous loss of water permeability and salt rejection,
 270 potentially due to the aggregation of the AgNPs. The overlaying of AgNPs on each
 271 other (see TEM of TFC-Ag100 in Figure 2) not only increases the effective water
 272 transport path but also deteriorates the polyamide rejection layer.

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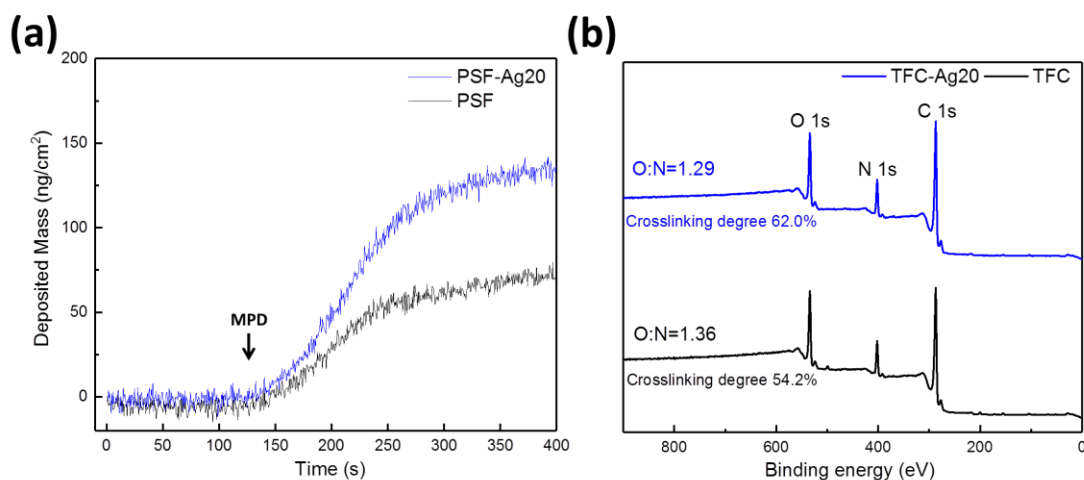
274 We further compared the permeability and selectivity of the TFC-Ag20 membrane to
 275 TFC RO membranes reported in the recent literature (e.g., TFN,^{7, 8} freestanding
 276 polyamide⁵¹ and layer-by-layer polyamide membranes⁵²) as well as representative
 277 commercial RO membranes. In Figure 3b, the TFC-Ag20 membrane exhibited both
 278 excellent water permeability and salt selectivity, exceeding the conventional
 279 permeability-selectivity trade-off.



280

281 Figure 3. (a) Separation performance results of the control and TFC-Ag membranes at an
 282 applied pressure of 20 bar using 2000 ppm NaCl solution, (b) tradeoff of membrane water
 283 permeability and selectivity (water over NaCl) based on commercial membranes and
 284 literature data.^{7, 8, 51, 52} The salt rejection tests for all commercial membranes and literature

285 data were carried out with 2000 ppm NaCl solution and their water permeability A values (L
286 $m^{-2}h^{-1}bar^{-1}$) have been normalized by pressure.
287



288

289 Figure 4. (a) QCM-D sorption test of MPD monomer (1 wt%) on the control and PSF-Ag20
290 substrates coated quartz crystal sensor, (b) XPS spectra of the control TFC and TFC-Ag20
291 membranes.

292

293

294 **AgNPs-induced nanochannels enhance the rejection of organic**

295 **micropollutants.** Figure 5a presents the rejection of various solutes by the control

296 and the TFC-Ag20 membrane. In this comparison, we included a salt (NaCl), a set of

297 neutral hydrophilic solutes (boron, ethylene glycol, diethylene glycol, and

298 pentaerythritol, norfloxacin, and ofloxacin), and a hydrophobic compound

299 propylparaben. In all cases, the TFC-Ag20 membrane had better rejection compared

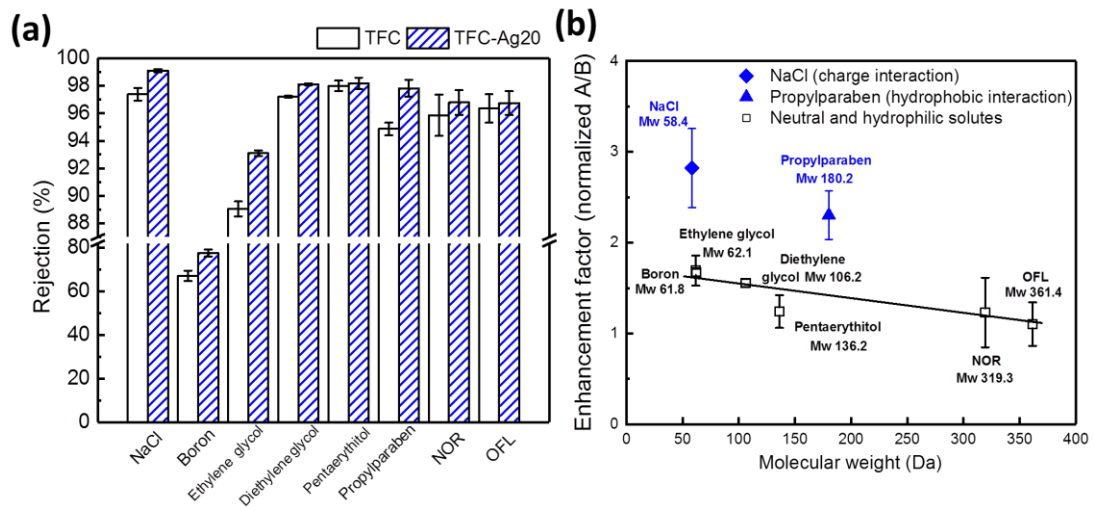
300 to the control TFC membrane, which is partially attributed to its enhanced

301 crosslinking degree (Figure 4b) and thus greater size exclusion effect. To gain a better

302 understanding of the underlining mechanisms, we analyzed the selectivity

303 enhancement factor for each compound by normalizing the respective A/B value of

304 the TFC-Ag20 membrane against that of the control membrane.³⁹ A higher normalized
305 *A/B* value indicates greater improvement in selectivity for the AgNPs-loaded
306 TFC-Ag20 membrane. Figure 5b, the data points for the neutral hydrophilic solutes
307 collapsed into a single trend line, which reveals the contribution of the improved size
308 exclusion effect.⁵³ In general, the enhancement factor of the TFC-Ag20 membrane
309 systematically decreased with the increased molecular weight. This is reasonable;
310 when the size of the solute is large than the pores of both membranes, the selectivity
311 enhancement factor would theoretically approach to unity (i.e., no enhancement) since
312 both membranes would completely reject the solute. Interestingly, the salt (NaCl) and
313 the hydrophobic compound (propylparaben) showed greater enhancement in
314 selectivity compared to the trend line that incorporated the effect of size exclusion.
315 This result suggests the involvement of additional mechanisms. The charged NaCl
316 benefits from the additional Donnan exclusion effect with the presence of the
317 negatively charged AgNPs (Table S1). On the other hand, the suppressed hydrophobic
318 interaction for the propylparaben (thanks to the hydrophilic nature of AgNPs), in
319 addition to its enhanced size exclusion effect, explains the greatly improved
320 selectivity of the TFC-Ag20 membrane against this hydrophobic compound.³⁹
321



322

323 Figure 5 (a) Membrane rejection results of the control TFC and TFC-Ag20 membranes on
 324 charged (NaCl), neutral (boron and diethylene glycol), hydrophobic (propylparaben) and
 325 hydrophilic solutes (OFL and NOR), (b) the correlation between the molecular weight of
 326 these various solutes and the enhancement factor of membrane selectivity (normalized A/B).

327

328 **IMPLICATIONS AND PERSPECTIVES**

329 In this study, we revealed that hydrophilic AgNPs induced the formation of
330 nanochannels in TFN membranes, which significantly enhanced membrane water
331 permeability. The size of the nanochannels (2-3 nm) was much narrower than the
332 characteristic Debye length (>10 nm), leading to enhanced salt rejection due to the
333 Donnan exclusion effect. The incorporating of AgNPs also increased the crosslinking
334 degree of the resulting polyamide membrane for further enhanced membrane rejection
335 and selectivity. In addition, we revealed that these nanochannels could greatly
336 improve the removal of a hydrophobic endocrine disrupting chemical (EDC,
337 propylparaben), thanks to the weakened hydrophobic interaction. Due to their
338 significant health implications yet relatively poor removal of RO membranes, EDCs
339 presents a critical challenge in the context of water reuse.³ The current study may
340 potentially provide a rational approach to design membranes for enhanced removal of
341 EDCs as well as other hydrophobic micropollutants (e.g., pesticides and herbicides).

342

343 Although the current study focused on the use of AgNPs, the approach can be
344 potentially applied to other types of nanoparticles. Table 1 summarizes recently
345 published results for TFN membranes incorporating solid nanofillers. We categorize
346 these TFN membranes by the phase that nanofillers are introduced (water vs. organic),
347 the type of nanofillers, and their hydrophilicity/hydrophobicity. Incorporation of
348 hydrophobic nanofillers during interfacial polymerization, regardless in the water or

349 organic phase, appeared to decrease the water permeability;^{27 28} the solid nanofillers
 350 without nanochannels around them would block the water transport and increase the
 351 membrane resistance. In contrast, the incorporation of hydrophilic nanofillers in water
 352 phase could generally result in higher water flux, which is consistent with the
 353 nanochannel formation. The corresponding salt rejection can either increase or
 354 decrease; the latter may be attributed to agglomeration of nanofillers.³² In this regard,
 355 preloading nanofillers (e.g., by *in situ* generation of AgNPs in this study) can provide
 356 a more uniform particle loading (Figure 2) and thereby provide enhanced solute
 357 rejection. Finally, loading hydrophilic nanofillers in organic phase results in less
 358 consistent effect, with several studies^{25, 54, 55} reporting reduced water permeability.
 359 Future studies need to further investigate the underlying mechanisms governing the
 360 behavior of these membranes.

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Table 1. Comparison of recent solid nanomaterials incorporated TFN membranes.

Phase	Nanofiller	Hydrophilicity	Water flux	Salt rejection	Published year and Ref.	Remark
Water	AgNPs	Hydrophilic	Pw ↑ by 170%	NaCl rejection ↑	This work	Preloaded on PSF
	AgNPs		Pw ↑ by 110%	MgSO ₄ and NaCl rejection ↑	2017 ²³	Preloaded on PSF
	Silica		Pw ↑ by 130%	NaCl rejection ↑	2009 ¹⁴	IP
	AgNPs		Pw ↑ by 82%	MgSO ₄ and NaCl rejection ↓	2012 ²⁴	IP
	Silica		Pw ↑ 150	NaSO ₄ rejection ↗ ↘	2013 ⁵⁶	IP
	TiO ₂		Pw ↑ by 70%	NaCl rejection ↓	2013 ⁵⁵	IP
	Na ⁺ -CQDs ^a		Pw ↑ by 47%	NaCl rejection unchanged	2019 ²¹	IP
	CuNPs		Hydrophobic	Pw ↓ 50%	NaCl rejection unchanged	2018 ²⁷
Organic	AgNPs	Hydrophilic	Pw slightly ↓	MgSO ₄ rejection ↑; No change in NaCl rejection	2007 ²⁵	IP
	Silica		Pw ↑ by 30%		2012 ⁸	IP
	TiO ₂		Pw slightly ↓	MgSO ₄ rejection ↑	2007 ⁵⁴	IP
	TiO ₂		Pw slightly ↓	NaCl rejection ↑	2013 ⁵⁵	IP
	Fluorinated silica	Hydrophobic	P _w ↓ 35%	NaCl rejection ↗ ↘	2017 ²⁸	IP

364 **ASSOCIATED CONTENT**

365 **Supporting Information**

366 The Supporting Information is available free of charge on the ACS Publications
367 website at DOI:

368 S1. SEM micrographs, silver mass density, contact angles and permeability of
369 the control and AgNPs modified PSF substrates; S2. Size distributions of the
370 AgNPs induced nanochannels; S3. Physiochemical properties of TFC
371 membranes; S4. Zeta potential of AgNPs; S5. Calculation of Debye length; S6.
372 Calculation of membrane crosslinking degree; S7. Membrane rejection of neutral
373 hydrophilic solutes; S8. Membrane fouling behavior.

374

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378 *Notes*

379 The authors declare no completing financial interest.

380

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384

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