Phase diagram and spin-glass phenomena in electron-doped La$_{1-x}$Hf$_x$MnO$_3$ (0.05 ≤ x ≤ 0.3) manganite oxides

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The effects of tetravalent hafnium doping on the structural, transport, and magnetic properties of polycrystalline La$_{1-x}$Hf$_x$MnO$_3$ (LHMO) (0.05 ≤ x ≤ 0.3) were investigated systematically. LHMO exhibited a typical colossal magnetoresistance effect via the double-exchange between Mn$^{2+}$ and Mn$^{3+}$ ions, instead of that between Mn$^{3+}$ and Mn$^{4+}$ ions in hole-doped manganites. A phase diagram was obtained for the first time through magnetization and resistance measurements in a broad temperature range. As the Hf concentration varied from x = 0.05 to 0.3, the Curie point and metal-to-insulator transition temperature increased significantly, whereas the magnetization and resistivity decreased remarkably. An abnormal enhancement of the magnetization was observed at about 42 K. It was further confirmed that a second magnetic phase MnO$_2$ in LHMO gives rise to such a phenomenon. The possible causes are discussed in detail. The dynamic magnetic properties of LHMO, including relaxation and aging processes, were studied, demonstrating a spin-glass state at low temperature accompanied by a ferromagnetic phase. © 2011 American Institute of Physics. [doi:10.1063/1.3666056]

I. INTRODUCTION

One of the current interests in condensed matter physics is the investigation of strong electronic correlation in transition metal oxides. In doped manganites, the interactions among charge, orbital, spin, and lattice result in a variety of intriguing phenomena, such as colossal magnetoresistance (CMR), charge-orbit-ordering, and high-temperature superconductivity. The parent compound LaMnO$_3$ is a charge-transfer insulator with trivalent manganese in different layers that couple themselves anti-ferromagnetically through a superexchange (SE) mechanism. But within a layer, these Mn$^{3+}$ ions are coupled ferromagnetically. Usually, doping divalent elements in manganites, such as Sr, Ca, and Ba, can introduce holes in the g band via the partial substitution of a proportionate amount of Mn$^{3+}$ ions (with the electronic configuration $t_{2g}^3e_g^1$ and total spin $S=2$) with Mn$^{4+}$ ions ($t_{2g}^5$, single localized spin $S=3/2$). The holes can be permitted charge transfer in the $e_g$ orbital, which has a strong hybridization with the 2p state of a neighboring oxygen ion. As a result, due to the intraatomic Hund’s rule, hopping of the $e_g$ electrons can induce ferromagnetic coupling between Mn$^{3+}$ and Mn$^{4+}$ ions. It is responsible for the ferromagnetism and the associated metallic conductivity in the doped manganite oxides. Meanwhile, the reduction in the disorder of the manganese spins will cause a decrease in resistivity and lead to a metal-to-insulator transition (MIT) around the Curie temperature ($T_C$). The double-exchange (DE) mechanism and strong electron-phonon interaction arising from Jahn-Teller distortion were introduced in order to fully clarify the origin of the CMR effect. Following the study of Das et al., many researchers attempted to substitute tetravalent Ce$^{4+}$, Te$^{4+}$, Sn$^{4+}$, and Sb$^{5+}$ ions at the La site in order to gain electron doped manganite oxides. Their results revealed that the physical and chemical properties are strongly dependent on the doped elements and doping level. It is generally believed that these so-called electron-doped manganites exhibit the CMR effect via the DE between Mn$^{3+}$ and Mn$^{2+}$ ions, instead of DE between Mn$^{3+}$ and Mn$^{4+}$ ions as in hole-doped compounds. This creates the possibility of fabricating all manganese $p$-$n$ junctions and spintronic devices in the near future. However, as many tetravalent elements can show multivalent states, arguments can be raised as to whether it is intrinsically electron-type or hole-type for the conductive mechanism in a tetravalent ion doped LaMnO$_3$ system. Unlike other tetravalent elements, Hf typically shows a single tetravalent state or zero. This makes studies of the electronic structure of the La$_{1-x}$Hf$_x$MnO$_3$ system relatively simple and more reliable. On the other hand, the radius of the Hf$^{4+}$ ion lies intermediately among commonly used dopants [$\text{Sn}^{4+}(0.081 \text{ nm}) < \text{Hf}^{4+}(0.083 \text{ nm}) < \text{Ce}^{4+}(0.097 \text{ nm}) < \text{La}^{3+}(0.106 \text{ nm})$]. It is expected that the substitution of La$^{3+}$ by Hf$^{4+}$ might form a similar electron-doped manganese oxide like the others. Recently, Hf doped manganese [La$_{1-x}$Hf$_x$MnO$_3$ (LHMO)] was reported by our group. Our previous results from x-ray photoemission spectroscopy (XPS) reveal that the LHMO compound is in a mixed valence state of Mn$^{2+}$ and Mn$^{3+}$, and the magnetic-field dependence of Hall resistivity further confirmed that the carriers in LHMO are electrons. All the results imply the electron conduction mechanism in LHMO. Although the properties of epitaxially grown...
LHMO thin films have been studied, there has been no systematic investigation of the physical properties of LHMO bulk materials with different doping levels. In this paper, we investigate the effects of tetravalent Hf doping on the structural, transport, and magnetic properties of polycrystalline LHMO samples. The phase diagram was obtained for the first time through magnetization and resistance measurements. An abnormal enhancement of the magnetization induced by the impurity phase is discussed in detail.

II. SAMPLE PREPARATION AND EXPERIMENTAL DETAILS

Bulk polycrystalline samples of $\text{La}_{1-x}\text{Hf}_x\text{MnO}_3$ (0.05 $\leq x \leq 0.3$) were prepared via the standard solid state reaction method. Stoichiometric amounts of $\text{La}_2\text{O}_3$, $\text{Hf}_2\text{O}_3$, and $\text{MnO}_2$ high-purity powders were mixed together sufficiently and preheated to 900 °C for 12 h. Then, it was pressed into pellets and sintered in air at 1200 °C for 96 h with intermediate grindings and cooled down to room temperature slowly. A second sintering by repeating the second step was carried out in order to improve the purity of samples. The structure of the LHMO samples was examined via powder x-ray diffraction (XRD) measurements using Cu Kα radiation. The dc magnetization measurements, as a function of temperature (10–300 K) and applied magnetic field (from $-5$ to $+5$ T), were carried out. The electrical transport properties were measured via the standard four-probe method. Electric contacts were made using silver paste with a contacting resistance below $0.05 \, \Omega$ at room temperature. The dc magnetization and resistance of the LHMO samples were measured using a Physical Property Measurement System (PPMS-9 T, Quantum Design). The data acquisition was done in dc mode, which measures the voltage under an alternative dc current, and the sample resistivity is obtained by averaging these signals at each temperature. In this way, the contacting thermal power is naturally removed. The temperature stabilization was better than 0.1%, and the resolution of the voltmeter was better than 10 nV.

III. RESULTS AND DISCUSSIONS

A. Structural analysis

Figure 1(a) shows the XRD $\theta$–$2\theta$ scanning curves of LHMO with different Hf doping levels. Almost all main peaks from the samples can be indexed by a rhombohedral structure. As we can find in Fig. 1(a), the peaks from the secondary phase cannot be observed when the doping level increases from 0.05 to 0.3. It was further confirmed as the unreacted $\text{MnO}_2$ phase, which is probably due to the difference in ionic radius between $\text{La}^{3+}$ (0.106 nm) and $\text{Hf}^{4+}$ (0.083 nm). We fitted the XRD data using the software POWDER-X and obtained the lattice parameters of all the LHMO samples. As shown in Fig. 1(b), the doping levels dependent on the $c$-axis and $a$-axis lattice constant of LHMO were presented. It is obvious that the $a$-axis lattice constant shrinks, whereas the $c$-axis expands monotonically with the Hf concentration, which clearly indicates that Hf ions have been successfully substituted into the La sites. A smaller $c$-axis and larger $a$-axis constant will lead to a decrease in the Mn-O bond length and an increase of the Mn-O-Mn bond angle. Therefore, the $\text{MnO}_6$ octahedral will be distorted. Consequently, both DE and SE interactions will compete more strongly, with dependence on the structural distortion, giving rise to a magnetically disordered state. It can affect the magnetic and electrical transport properties of LHMO dramatically.

B. Resistivity, magnetization, and phase diagram of LHMO (0.05 $\leq x \leq 0.3$)

The magnetic and electrical transport properties of LHMO with different doping levels were investigated. Figure 2(a) shows the temperature dependence of the dc magnetization of LHMO ($x=0.05$, 0.1, 0.2, and 0.3) under a magnetic field of 200 Oe measured in both zero-field-cooling (ZFC) and field-cooling (FC) modes. All the samples exhibit a para-magnetic (PM) to ferromagnetic (FM) transition behavior. Several explicit effects on the magnetization of LHMO induced by doping with Hf were observed. Firstly, the slope of the magnetization around $T_C$ becomes more pronounced with higher Hf concentrations. Secondly, as the Hf content increases, there is a significant drop in the total magnetization at low temperature and an increase of $T_C$. The $T_C$, defined as the peak temperatures in the curves of $dM/dT$ versus temperature, are 170, 180, 230, and 225 K for $x=0.05$, 0.1, 0.2, and 0.3, respectively. The results might be caused by the competition between the DE and the core spin interaction, which leads...
to the canting of the core spins as the doping level increases. Thirdly, a bifurcation occurs between the ZFC and FC curves at low temperatures when LHMO is measured in a 200 Oe field. This is a typical spin-glass behavior and can be observed in either hole-doped manganites like La_{1-x}Sr_{x}MnO_{3} and La_{1-x}Ca_{x}MnO_{3} (Refs. 7–11) or other electron-doped manganites like La_{1-x}Ce_{x}MnO_{3} and La_{1-x}Te_{x}MnO_{3}.\textsuperscript{15–21} Moreover, an abnormal enhancement of the magnetization was found at about 40 K when x ∼ 0.1. It is suggested that an impurity phase mixed in LHMO compounds might give rise to such a phenomenon (a detailed discussion can be found in Sec. III C).

Figure 2(b) displays the temperature dependence of the resistivity (ρ) for the LHMO (0.05 ≤ x ≤ 0.3) without magnetic fields. For x = 0.05 and 0.1, the LHMO shows an insulator behavior (dρ/dT < 0) at all temperatures. LHMO with x = 0.2 and 0.3 presents a MIT behavior with decreasing temperature. The resistivity of LHMO decreases with increasing doping level x as the result of increasing carriers. We measured the temperature dependence of the resistivity for LHMO (0.2) under various fields of 0, 0.02, 0.1, 0.25, 1, 2, 3, and 5 T, as shown in Fig. 3. At zero magnetic field, the ρ-T curve shows two independent peaks: a sharp peak at higher temperature (T_{R1} ∼ 232 K) and a broad, larger peak at lower temperature (T_{R2} ∼ 210 K). Such a shoulderlike feature had also been observed for oxygenated La_{2/3}Ba_{1/3}MnO_{3} samples\textsuperscript{13} and La_{0.7}Ce_{0.3}MnO_{3} samples.\textsuperscript{14} Both of them indicated that the first-order nature of this transition in the vicinity of the double peaks is caused by oxygen overdoping. With increasing magnetic field H, ρ drops dramatically in all temperature ranges. The sharp peak at T_{R1} is gradually suppressed and shifts to higher temperatures with increasing magnetic fields, whereas the position of the peak at T_{R2} has never moved. The inset of Fig. 3 shows the field dependence of T_{R1}. T_{R1} increases from 232 to 280 K when H increases from 0 to 5 T, demonstrating a typical characteristic of CMR materials. It is evident that T_{R1} is the real MIT temperature. The temperature dependence of MR (defined as MR = \left[\frac{R(0) - R(H)}{R(0)}\right] × 100\%) for different magnetic fields is also shown in Fig. 3. The MR versus T curves shows a gradual growth peak at T_{R1} with increasing H. The MR values of LHMO(0.2) are 0.59%, 1.92%, 5.79%, 17.96%, 29.66%, 38.91%, and 50.9% under H = 0.02, 0.1, 0.25, 1, 2, 3, and 5 T measured at T_{R1}, respectively. The same measurements were carried out in LHMO (0.3) samples, and the MIT temperature of LHMO (0.3) was about 236 K. The Curie temperature T_{C} from M-T curves corresponds with the
transition temperatures $T_p$ from $\rho$-$T$ curves for LHMO with high doping levels. Based on these data, a phase diagram of LHMO as a function of $T$ and Hf content within $x = 0.05$–0.3 could be obtained, as shown in Fig. 2(c). The $T_C$ of the undoped LaMnO$_3$ compound is about 160 K, as obtained from Schiffer et al. The solid line in the figure is a guide for the eye in order to point out the approximate boundary between the spin-disordered PM state and the spin-ordered FM state. As one can see, the hatched area indicates the approximate boundary of the ground states of insulating and metallic conduction modes.

However, one might argue whether LHMO is intrinsically electron-type or hole-type for the conductive mechanism. In our previous works, the XPS measurements revealed that the binding energies of the Hf 4f-spectrum in LHMO were found to be about 2 eV greater than that of Hf metal and almost identical to the 4f binding energies of HfO$_2$, which suggests that the Hf$^{4+}$ ion is indeed in the tetra-valent state in LHMO. At the same time, the shape of the Mn 2p doublet and the value of the Mn 3s exchange splitting provide clear evidence of the presence of Mn$^{2+}$ ions. Thus we conclude that our LHMO film is in a mixed valence state $\text{Mn}^{2+}$ and $\text{Mn}^{3+}$, implying an electron-doped conduction mechanism. Hall effect measurements were also carried out in LHMO. In the strongly correlated systems, although the Hall effect is complex because of the anomalous Hall effect, it is possible to identify the carrier type in terms of the ordinary Hall effect. In ferromagnets, the magnetoresistance contains an additional term that depends on the magnetization,

$$\rho_H = R_0 B + \mu_0 R_S M,$$

$$B = \mu_0 [H + (1 - N) M],$$

where $R_0$ is the ordinary Hall coefficient, $R_S$ is the anomalous Hall coefficient, $M$ is the magnetization, $N$ is the demagnetization factor, and $\mu_0$ is the vacuum permeability. To separate the two terms, we measured $M(H)$ on the same sample, and $M$ is saturated when $H > 2$ T at 170 K. The second term will not change the slope of $\rho_H(H)$ at high fields. The Hall resistance fit linearly at $H > 2$ T. The curves for $\rho_H(H)$ at both temperatures exhibit a negative high-field slope, similar to what was reported by Yang et al., implying that the charge carriers in LHMO are electrons.

In addition, one might also doubt that the CMR behaviors observed in our experiments are the result of the creation of oxygen interstitial sites or vacancies at the La sites in our samples. The $T_p$ of LHMO ($\sim$230 K) is much higher and the resistivity of LHMO ($\sim 10^{-3}$ to $10^{-4}$ $\Omega$ cm) is several orders of magnitude lower than those of LaMnO$_3$ and $x$=0.2 LHMO ($T_p$ $\sim$ 110 K and $\rho$ $\sim 10^3$ $\Omega$ cm). Therefore, we can exclude the effects of LaMnO$_3$ or the oxygen deficiency or La-site deficiency on the magnetic behaviors. In the generic hole-doped manganite, the CMR effect can be well understood by the DE effects between the Mn$^{3+}$ and Mn$^{4+}$ ions. If the Hf-doped manganites, the rare-earth ion is partially replaced by tetravalent Hf$^{4+}$ ions, and the corresponding amount of manganese will be converted into a divalent state. The mixed-valence state of manganese ions in LHMO is Mn$^{2+}$-$\text{Mn}^{3+}$ instead of Mn$^{3+}$-$\text{Mn}^{4+}$. There are some differences between the hole-doped and electron-doped manganites, such as the electronic configuration being $(t^2_2 e^4_g)$, $(t^2_2 e^1_g)$, and $(t^2_2)$, and the sums of the spin are $S = 5/2$, 2, and 3/2 for Mn$^{2+}$, Mn$^{3+}$, and Mn$^{4+}$, respectively. The 3d state densities of Mn$^{4+}$ and Mn$^{2+}$ are greatly different from each other. All of these lead to the differences between the wavefunction $\Psi$ and the Hamiltonian $H$ when calculating the function of the DE model. Thus, it might be necessary to make some modifications to the DE mechanism, which is well suited for the hole-doped manganites, when it is applied to the electron-doped manganites.

**C. Observation of secondary phase in LHMO compounds**

In Fig. 2(a), an abnormal enhancement of the magnetization was observed at about 40 K on the LHMO samples when $x \geq 0.1$. Similar results were also reported on other electron-doped manganites, such as La$_{1-x}$Te$_x$MnO$_3$ and La$_{1-x}$Ce$_x$MnO$_3$. However, the origin of the enhanced magnetization at low temperatures has not been well understood or discussed. The enhanced magnetization at low temperature is usually attributed to the following things: (i) temperature-induced structure transformation, (ii) spin reorientation in magnetization, and/or (iii) a magnetic impurity phase mixed in manganites. We measured the magnetization of LHMO with $x = 0.2$ (LHMO(0.2)) under fields of 200 and 1000 Oe in the range of 10–60 K with a step of 1 K, as shown in Figs. 4(a) and 4(b). The measurements were carried out under fields of (a) 200 Oe and (b) 1000 Oe in the ZFC and FC modes. An obvious bifurcation occurs between two measured modes. (c) The temperature dependence of the magnetization for MnO$_2$ powder in an applied field of 200 Oe. Note that an anomalous enhancement of the magnetization occurs at 42 K in both LHMO and impurity phase MnO$_2$. 

![FIG. 4](https://example.com/fig4.png)
out in ZFC and FC modes. The second transition happened exactly at 42 K, as determined by the peak of $dM/dT$ versus temperature curves. Despite the fact that LHMO(0.2) samples entered the spin-glass state at about 150 K (as shown in Fig. 2(a)), the result shows a larger and significant bifurcation between ZFC and FC curves when the temperature is below 42 K. In Figs. 4(a) and 4(b), it can be found that with increasing temperature, the deviation between ZFC and FC magnetization ($\Delta M$) gradually decreases. With an increase in the external field from 200 to 1000 Oe, $\Delta M$ also decreases at the same measuring temperature. This is because the spin-glass state is not at equilibrium with an energy higher than that of the long-range FM equilibrium state. The transition from spin-glass state to FM state should overcome the energy barrier, which will decline gradually with an increase of temperature and magnetic field.

The impurity phases were found in the electron-doped manganites commonly. By intensively analyzing the XRD data of La$_{0.7}$Ce$_{0.3}$MnO$_3$, Mitra et al. identified the impurity phase in the La$_{0.7}$Ce$_{0.3}$MnO$_3$ bulk samples as unreacted CeO$_2$. Recently, Chou et al. further examined the microscopic morphology and composition distribution of La$_{0.7}$Ce$_{0.3}$MnO$_3$ and La$_{0.8}$Te$_{0.2}$MnO$_3$ bulks. Besides their own phases, the former contains a multiphase of La$_{0.85}$Ce$_{0.017}$MnO$_3$ and CeO$_2$, and Mn-O, and the latter contains a lanthanum deficient La$_{0.8}$MnO$_{3-x}$. However, it is hard to understand the abnormal enhancement of the magnetization at low temperature when mixed with nonmagnetic impurity phases of CeO$_2$ or other perovskite structures. Gebhardt et al. have observed a second transition of the magnetization in La$_{1-x}$Ce$_x$MnO$_3$ compounds around 40 K, which was attributed to the antiferromagnetic transition arising from unreacted MnO$_2$. Figure 4(c) presents the magnetization versus temperature scan of MnO$_2$ powder under 200 Oe field in both ZFC and FC modes. The Néel temperature $T_N$ ($\sim$95 K) of MnO$_2$ powder is indicated in Fig. 4(c). An enhanced magnetization occurs in MnO$_2$ at the same temperature of 42 K as in the LHMO(0.2) samples. The magnetization of MnO$_2$ increases suddenly by 40% when the temperature decreases from 45 to 35 K. The bifurcation between ZFC and FC becomes very distinct below 42 K, which coincides with the results found for LHMO(0.2). Therefore, the impurity phase of MnO$_2$ could provide a big contribution to the total magnetization of LHMO bulks at low temperature.

Figure 5(a) shows the temperature dependence of the magnetization for LHMO(0.2) at magnetic fields of 0.02, 0.1, 1, 2, and 5 T. The magnetization was measured in FC mode. (b) The field dependence of the magnetization for LHMO(0.2) at constant temperatures of 10, 45, 150, and 300 K. The inset of (b) shows the expanded view of the LHMO(0.2) magnetization curves around the saturation magnetization.
metals under standard conditions. They have two electrons in their valence shell, so their energetically preferred state of achieving a filled electron shell is to lose two electrons to form doubly charged positive ions. Therefore, it is thought that these elements could react with MnO₂ or MnCO₃ compounds sufficiently during the solid state reaction, and little secondary phase can be found in hole-doped manganites. However, the tetravalent elements, which are used to form electron-doped manganites, have four electrons in the external shell and are relatively chemically stable. Doping tetravalent elements in the manganites seems harder than doing so with divalent elements. We think that might be the reason for the impurity phase contained in electron-doped manganites commonly. Further investigations of the mechanism of Hf doped manganites are under way. We are planning to do some other measurements to test whether any other impurity phases are mixed in LHMO.

In order to get further information on the low temperature magnetic properties of the LHMO system, the dynamic magnetic properties have been investigated via dc magnetization measurement, including relaxation and aging processes. The samples were first cooled down from a reference temperature in the PM state to the measuring temperature and kept for a waiting time \( t_w \). Thereafter, a dc magnetic field was applied and the \( M(t) \) was recorded. Figure 6 displays the time dependence of \( M(t) \) measured in a field of 200 Oe at 10, 60, and 300 K after different waiting times \( t_w \). After a magnetic field was applied, the increasing relaxation of the magnetization could be observed at 10 and 60 K, like in the FM phase. After the sample experiences a transition from FM to PM, the magnetization decreases with time due to the magnetic relaxation, as presented in Fig. 6(c). It is clearly found that the magnetization is strongly dependent on \( t_w \), which demonstrates that the system possesses the aging process. Among the various physical models proposed to discuss the temperature dependence of magnetization, one of the most popular is a stretched exponential form as in

\[
M(t) = M_0 - M_r \exp \left[ - \left( \frac{t}{t_w} \right)^{1-n} \right], \tag{3}
\]

where \( M_r \) relates to the intrinsic FM component and \( M_r \) to the glassy component mainly contributing to the relaxation effects observed. We fitted \( M(t)/M_0 \) with Eq. (3). The fitting parameters are shown in Table I. \( M_r/M_0 \) depends on the temperature and decreases with increasing temperature. The waiting time \( t_w \) shows a significant effect on the magnetic relaxation, revealing an intrinsic aging process. This is a manifestation of the nonequilibrium nature of the spin-glass state. As we all know, to form the spin-glass state, two important factors must exist: disorder of sites or bonds, and completion between different magnetic phase interactions. Owing to the random substitution of \( \text{La}^{3+} \) with \( \text{Hf}^{4+} \), there might exist disordered distributions of \( \text{Mn}^{2+} \) and \( \text{Mn}^{3+} \) ions in LHMO compounds. This could lead to a random distribution of the magnetic constants \( J_{\text{FM}} \) and \( J_{\text{AFM}} \), as well as make magnetic frustration and FM clusters easy to form. Meanwhile, the substitution of different sizes of ions will result in lattice distortions, which might influence the FM DE and AFM SE interactions differently. As mentioned above, replacing \( \text{La} \) with \( \text{Hf} \) would lead to a decrease in \( \text{Mn-O} \) bond length and an increase of the \( \text{Mn-O-Mn} \) bond angle. Thus the enhanced lattice distortion would weaken the DE and favor SE interactions. This competition is sufficient to suppress long-range FM order and frustrated interaction among magnetic moments. DE and SE interactions would compete more strongly, with dependence on the structural distortion, giving rise to a magnetically disordered state. Therefore, we conclude that the coexistence and completion between the FM and AFM interactions, along with the randomness, would lead to the spin-glass state in LHMO compounds.

**IV. CONCLUSION**

The effects of tetravalent Hf doping in manganite oxides on the structural, magnetic, and transport properties are systematically studied in this paper. By measuring \( M-T \) and \( \rho-T \) curves of LHMO with different doping levels, we obtained the phase diagram of LHMO compounds \((0.05 \leq x \leq 0.3)\) for the first time. It was found that the magnetic and electrical transport properties were strongly dependent on the Hf concentration. An abnormal enhancement of the magnetization in LHMO with \( x \geq 0.1 \) was observed at about 42 K and was attributed to the contribution of an unreacted MnO₂ impurity phase. There is coexistence of ferromagnetic and spin-glass states in the LHMO compounds at low temperature. Dynamic magnetic measurements in LHMO were carried out, including relaxation and aging studies, and further confirmed the nature of spin-glass behavior. Although it is fully
understood that the observed results call for more experiments and theoretical work, our results should be meaningful evidence of the electron-doped manganite family. Our study highlights the importance of double-exchange interactions and electronic correlation effects in understanding the occurrence of a ferromagnetic state accompanied with a spin-glass state in electron-doped manganites.

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