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A Bilayer 2D-WS₂/Organic-Based Heterojunction for High-Performance Photodetectors

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Abstract: Two-dimensional (2D) tungsten disulfide (WS₂) has inspired great efforts in optoelectronics, such as in solar cells, light-emitting diodes, and photodetectors. However, chemical vapor deposition (CVD) grown 2D WS₂ domains with the coexistence of a discontinuous single layer and multilayers are still not suitable for the fabrication of photodetectors on a large scale. An emerging field in the integration of organic materials with 2D materials offers the advantages of molecular diversity and flexibility to provide an exciting aspect on high-performance device applications. Herein, we fabricated a photodetector based on a 2D-WS₂/organic semiconductor materials (mixture of the (Poly-(N,N'-bis-4-butylphenyl)-N,N'-bisphenyl) benzidine and Phenyl-C61-butyric acid methyl ester (Poly-TPD/PCBM)) heterojunction. The application of Poly-TPD/PCBM organic blend film enhanced light absorption, electrically connected the isolated WS₂ domains, and promoted the separation of electron-hole pairs. The generated exciton could sufficiently diffuse to the interface of the WS₂ and the organic blend layers for efficient charge separation, where Poly-TPD was favorable for hole carrier transport and PCBM for electron transport to their respective electrodes. We show that the photodetector exhibited high responsivity, detectivity, and an on/off ratio of 0.1 A/W, 1.1 × 10¹¹ Jones, and 100, respectively. In addition, the photodetector showed a broad spectral response from 500 nm to 750 nm, with a peak external quantum efficiency (EQE) of 8%. Our work offers a facile solution-coating process combined with a CVD technique to prepare an inorganic/organic heterojunction photodetector with high performance on silicon substrate.

Keywords: 2D-WS₂; photodetector; organic semiconductor; responsivity

1. Introduction

Two-dimensional materials (2D), such as graphene, hexagonal boron nitride, transition metal dichalcogenides, tin sulfide, black phosphorus, ultrasmall bismuth quantum dots, and selenium nanoflakes, have become optically active semiconductors in biomedicine, ion detectors, and photodetectors (PDs). Photodetector-based graphene exhibits a large-response wavelength range and a high on/off ratio. Two-dimensional black phosphorus, with an anisotropic band structure, shows linear dichroism and anisotropic absorption. Photodetector-based monolayer black phosphorus exhibits excellent polarization sensitivity with a large bandwidth. Photoelectric properties of transition metal dichalcogenides (TMDs) have a remarkable change, as bulk transforms into a monolayer or multilayers, which have high optical absorption rates and suitable band gaps. Photodetector-based monolayer transition metal dichalcogenides (TMDs) show high quantum efficiency and low response times [1–12]. Among these 2D materials, tungsten disulfide (WS_2) layers with high mobility of $1000 \text{ cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1}$, a high optical absorption coefficient of ca. 10^6 cm^{-1} , and a band gap of 1.9 eV [13–16] are typical n-type 2D materials for electronic and optoelectronic device applications, making them compatible to combine with other materials to construct 2D van der Waals heterostructures.

Organic materials are suitable materials to vertically stack together with 2D WS_2 layers to form an energy-favorable inorganic/organic heterojunction, providing an effective interface to separate the electron-hole (e-h) pairs excited by incident photons. A significant advantage to employing organic materials is due to their facile solution processability [17–22]. Nowadays, it remains challenging to form uniform 2D WS_2 thin films on a large scale due to the coexistence of discontinuous and isolated single layers and multilayers [23–27]. Due to the excellent flexibility of organic materials, the surface of the isolated 2D WS_2 layers could be effectively smoothed out and interconnected together, filling pinholes and vacancies of the 2D layers underneath. Recently, research has been intensively carried out to combine 2D inorganic materials with organic materials to form energy-favorable junctions for efficient exciton dissociation and charge transport [28–34]. Two-dimensional CsPbBr_3 nanosheets have been assembled into flexible heterojunction films with phenyl-C61-butyric acid methyl ester (PCBM), and the 2D CsPbBr_3 /PCBM photodetectors exhibited an enhanced responsivity of 10.85 AW^{-1} and an ultrahigh detectivity of $3.0 \times 10^{13} \text{ Jones}$ [30]. The van der Waals heterojunction formed between pentacene and 2D MoS_2 has demonstrated an ultrafast charge transfer of 6.7 ps (especially the charge-separated state that lives for 5.1 ns) up to an order of magnitude longer than the pure 2D heterojunctions, suggesting the benefits of junction-stacking organic/2D materials [31]. Therefore, an emerging field in the integration of organic materials with 2D materials provides an exciting aspect on continuous 2D device fabrication. A hole transport material, Poly-(*N,N'*-bis-4-butylphenyl-*N,N'*-bisphenyl) benzidine (Poly-TPD), and an electron transport material, PCBM, are often used for hole and electron transporting layers in organic photodetectors and solar cells [35–37]. Two-dimensional WS_2 (conductive band edge at ca. -3.84 eV and valence band edge at ca. -5.82 eV for a monolayer) is a perfect match for the Poly-TPD/PCBM mixture (LUMO at -4.1 eV of PCBM to accept electrons and HOMO at -5.2 eV of Poly-TPD to accept holes), which is favorable for exciton dissociation [38–43].

The growth of monolayer WS_2 is always a big challenge, especially for the large-scale monolayer WS_2 . Herein, the large-scale monolayer WS_2 was successfully prepared using ZnO film as an auxiliary material on an SiO_2/Si substrate through chemical vapor deposition (CVD) for the first time. With a convenient solution spin-coating technique, the blend film of mixed Poly-TPD and PCBM at a weight ratio of 1:1 was deposited on top of the 2D WS_2 to form an ordered bilayer inorganic/organic diode. The 2D WS_2 thin films, obtained from ZnO-controlled growth through chemical vapor deposition (CVD), provided inert and solid surfaces. The van der Waals force interactions of 2D WS_2 and organic layers could allow for the planar growth of organic-based films with larger crystal grain sizes. In the meantime, organic layers could smooth out the surface of the 2D WS_2 layer for continuous device fabrication, which could substantially suppress dark currents. Thus, functionalized organic thin films and suitable architectures with 2D WS_2 must be well optimized in order to realize inorganic/organic-based PDs with high performance.

Overall, the 2D WS₂/Poly-TPD/PCBM PD obtained a broad spectral response from 500 nm to 750 nm with a peak external quantum efficiency (EQE) of 8% at a wavelength of 527 nm. When illuminated with a 450-nm excitation laser at a power intensity of 0.14 mW/cm², the PD showed a gate-tunable responsivity, a specific detectivity (D^*), and an on/off ratio of 0.1 A/W, 1.1×10^{11} Jones, and 100, respectively. Meanwhile, the responsivity dramatically increased with the laser excitation intensity, then saturated to 17 A/W when the drain voltage V_D was 10 V. As a result, the 2D inorganic/organic bilayer heterojunction was successfully demonstrated for a high-performance PD and could extend to other organics and 2D materials.

2. Materials and Methods

Fabrication and Characterizations

ZnO film was prepared on an SiO₂/Si substrate at a thickness of ca. 100 nm through DC reactive magnetron sputtering (Figure 1a) [44]. Metal zinc (DingWei, Dongguan, China) was used as a target. Argon (Ar) was used as the sputtering gas and oxygen as the reactant. Ar/O₂ (Ar 20 sccm, O₂ 60 sccm, 0.3 Pa) was introduced into the sputtering chamber. A negative bias voltage of −300 V was applied to keep glow discharge for 30 min. WS₂ film was prepared through a CVD method (Figure 1b) [45,46]. The WS₂ powders (Aladdin, Shanghai, China) were loaded in the center of the CVD system and heated to 1000 °C from room temperature over 30 min. An SiO₂/Si substrate with a layer of ZnO (ZnO/SiO₂/Si) was placed in the downstream region of the CVD. During the above process, the substrate was heated to ca. 700 °C, which was kept for 60 min. Ar/H₂ (H₂ 5%, 105 Pa, 35 sccm) was used as a carrier gas [47]. The spin-coating method was applied to prepare the Poly-TPD/PCBM (1:1) thin film on the top of the WS₂ thin film (Figure 1c). The Poly-TPD (Aladdin, Shanghai, China) and PCBM (Aladdin, Shanghai, China) were first dissolved in chlorobenzene solvent with a mass concentration of 10 mg/mL. The spin-casting processes were performed in a glove box. The blend layer of organic materials was spin-coated at 3000 rpm for 50 s, followed by annealing at 110 °C for 30 min. Cr–Au electrodes were e-beam-evaporated on top of organic thin films through a shadow mask with an exposed active area under a vacuum of 1.3×10^{-3} Pa (Figure 1d). Here, the channel width of the device was 20 μm, and the length was 100 μm. The final structure of the typical device, on top of an oxidized silicon wafer, was Au/Poly-TPD/PCBM/WS₂/Au. The doped silicon could be used as a back gate and SiO₂ as a gate dielectric. A schematic diagram of the prepared 2D WS₂/organic photodetector with the respective energy level alignments is shown in Figure 1d.

X-ray diffraction (XRD) was performed on a Thermo ARLXTRA (Geneva, Switzerland). X-ray photoelectron spectroscopy (XPS) spectra were performed on an Ulvac-Phi PHI5000 Versaprobe II (Kanagawa, Japan). UV-Vis absorption was performed on a Shimadzu UV-3600 (Kyoto, Japan). Field emission scanning electron microscopy (FESEM) was performed on a FEI Apreo S HiVac (Hillsboro, OR, USA). Raman spectra and photoluminescence (PL) were performed on a micro-Raman setup consisting of a 532-nm solid state laser, a Nikon inverted microscope (Ti eclipse, Tokyo, Japan), a long-pass edge filter (Semrock, New York, NY, USA), and a Raman spectrometer (Horiba, iHR320, Kyoto, Japan). An optical microscope image and a luminescence image were taken on a Jiangnan MV 3000 digital microscope (Nanjing, China). An electrical measurement was carried out on an Agilent 4200 SCS (CA, America) and a LakeShore TTPX (Columbus, OH, USA). The photodetection properties were examined with laser excitation wavelengths of 450 nm to 750 nm.

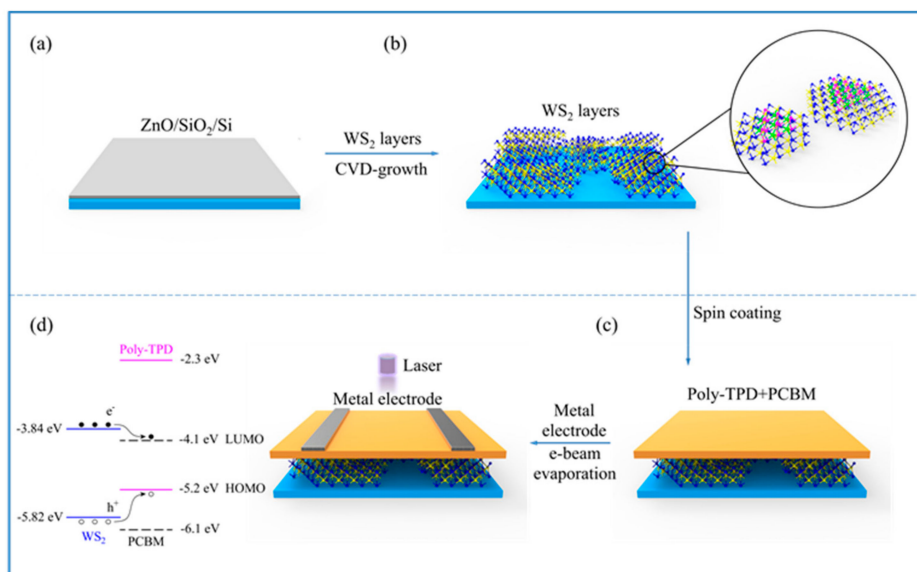


Figure 1. Illustration (color online) of the fabrication procedure of the photodetector based on the organic semiconductor/inorganic tungsten disulfide (WS_2) heterojunction. (a) The silicon substrate with ZnO and SiO_2 film ($\text{ZnO}/\text{SiO}_2/\text{Si}$). (b) Two-dimensional (2D) WS_2 growth on the surface of SiO_2/Si through a chemical vapor deposition (CVD) method. During the growth of the WS_2 , ZnO was removed from the SiO_2/Si . (c) (Poly-(N,N' -bis-4-butylphenyl- N,N' -bisphenyl) benzidine and Phenyl-C61-butyric acid methyl ester (Poly-TPD/PCBM) (1:1) organic film preparation by spin-coating on the surface of the 2D WS_2 . (d) Metal electrode evaporation on the surface of the Poly-TPD/PCBM film.

3. Results and Discussion

Figure 2a shows the XRD spectra of the $\text{ZnO}/\text{SiO}_2/\text{Si}$ substrate before and after the growth of the WS_2 film. The XRD data reveal that WS_2 film along with (002) direction was prepared on the Si substrate [48]. The full width at half maximum (FWHM) of the WS_2 XRD peak (002) was similar to that of the Si single crystal, indicating that a WS_2 film with a single crystalline with a large crystal grain size was obtained. There was no ZnO XRD peak that existed after complete 2D WS_2 growth (Figure 2a) [34], which was further confirmed by the XPS spectrum, since no Zn^{2+} signal (Zn $2\text{P}_{1/2}$ at 1021.75 eV and Zn $2\text{P}_{3/2}$ at 1044.7 eV) was detected (see Figure 2b) [49]. The detailed mechanism of 2D WS_2 controlled growth was discussed in our previous work [50]. The shape and scale of the monolayer WS_2 was affected by the atomic ratio of W/S. When the atomic ratio of W/S was less or larger than 1:2, the monolayer WS_2 would only grow into a small triangular shape. Therefore, it was critical to maintain the atomic ratio of W/S to as close to 1:2 as possible. Since ZnO whiskers can absorb extra W atoms to form a ZnWO_4 compound in a way that adjusts the atomic ratio of W/S back to 1:2, the WS_2 could be promoted further to grow into large-scale monolayer domains. We speculated the ZnWO_4 was sublimated and removed from the substrate during the subsequent growth stage.

The UV-Vis absorption spectrum of the WS_2 film is shown in Figure 2c. The A peak (located at 575–670 nm) and B peak (located at 527.7 nm) were due to the A and B exciton absorptions, respectively [51,52]. Due to spin-orbit coupling, the valence band split into two sub-bands (v_1 and v_2 , $v_1 < v_2$) at the K point of the Brillouin zone. Transitions from v_1 and v_2 to the minimum of the conduction band corresponded to the B exciton and A exciton, respectively [53]. The A peak consisted of two peaks at 584.1 nm ($\text{A}(\text{X})$) and 626.3 nm ($\text{A}(\text{X}^-)$). $\text{A}(\text{X})$ was due to the neutral exciton absorption, and $\text{A}(\text{X}^-)$ was due to the charged exciton absorption [54]. In addition, there was a broad peak I at the low-energy side of the A peak (>700 nm). The weak and broad peak I was ascribed to the indirect band gap transition of the multilayers/bulk [24]. Figure 2d shows the absorption spectrum of the organic Poly-TPD/PCBM (1:1) thin films, with two strong absorption peaks in the near-UV region (300–420 nm).

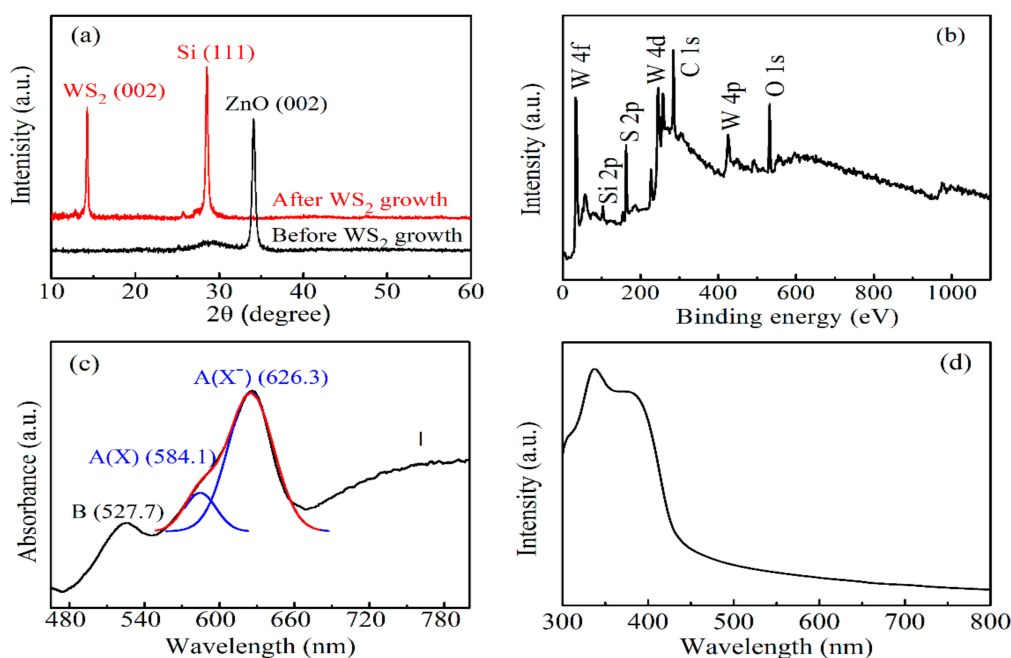


Figure 2. (a) The X-ray diffraction (XRD) spectra of the ZnO/SiO₂/Si substrate (black line) and WS₂ film (red line). (b) X-ray photoelectron spectroscopy (XPS) spectrum of the WS₂ film. UV-Vis absorption spectrum of the (c) WS₂ film and (d) Poly-TPD/PCBM (1:1).

Figure 3 shows typical FESEM images of WS₂ film grown on the silicon substrate via the CVD method. The white-colored area in Figure 3a,b is the SiO₂/Si substrate. The gray-colored areas with the hexagonal shape are WS₂ monolayers. The hexagonal morphology indicated that the WS₂ monolayer we prepared had a single crystalline with a large crystal grain size. The dark-colored areas with irregular morphology were due to the WS₂ multilayers. The small-sized multilayers were stacked on a monolayer with a large domain size. During WS₂ growth, hexagonal WS₂ monolayers merged together. As shown in Figure 3b, cracks in these hexagonal interfaces could be clearly viewed, and they could have formed during the CVD growth or cooling process. For the Poly-TPD/PCBM (1:1) blend films, the roughness was 0.5 nm in Figure 3c, indicating that Poly-TPD and PCBM were uniformly mixed together, which was required for the subsequent Au electrode deposition. If the surface of the organic blend film were not smooth, the deposited Au atoms could penetrate through the vacancies or voids of the 2D WS₂ materials, and the as-prepared devices would be very leaky. Therefore, the uniform feature of the blend film was crucial to a successful 2D WS₂/organic photodetector fabrication.

Typical Raman spectra of the WS₂ films at room temperature with a 532-nm laser excitation are shown in Figure 4a. The red and blue curves in Figure 4a approximately correspond to the areas circled with a red line and blue line in the inset of Figure 4b, respectively. The Raman peak of silicon (Si) at 520 cm⁻¹ was used to calibrate the Raman spectra of the as-prepared WS₂ thin films. Two typical Raman peaks at 351 cm⁻¹ (E_{2g}^1) and 417.3 cm⁻¹ (A_{1g}) were observed in the red curve [55]. The frequency difference between the two modes was 66.3 cm⁻¹. The peak intensity ratio of A_{1g}/E_{2g}^1 was 0.029. The narrow frequency difference, small peak intensity ratio, and weak A_{1g} intensity indicated that the WS₂ domain labeled with a red circle in Figure 4c was confirmed to be a monolayer [56]. For the blue curve in Figure 4a, the E_{2g}^1 redshifted to 348.6 cm⁻¹, and the A_{1g} blueshifted to 419.7 cm⁻¹. The frequency difference increased to 71.1 cm⁻¹. In addition, the intensity ratio increased to 0.25, indicating that the WS₂ domain circled with a blue line in the inset of Figure 4b had a multilayer feature [24]. Besides the E_{2g}^1 and A_{1g} peaks, we found a B_{2g}^1 peak as well in the blue curve. The phonon mode B_{2g}^1 was only active in the multilayers [57,58]. Thus, the WS₂ domain corresponding to the blue curve had multilayers. In addition, the absence of the B_{2g}^1 peak in the red curve indicates that the corresponding WS₂ domain was a monolayer. Figure 4b shows the typical photoluminescence (PL) spectra of the

WS₂ monolayer (red line) and multilayers (blue line) with a 532-nm laser excitation. Interestingly, the multilayer WS₂ showed a faint PL emission without obvious peaks, since multilayer WS₂ had indirect band gap characteristics. In contrast, monolayer WS₂ exhibited an intense and sharp PL spectrum with a typical peak at 619.2 nm, which indicated a direct band gap of the WS₂ monolayer [59]. Correspondingly, the PL mapping (Figure 4c) of the in situ optical microscopic view in Figure 4d clearly shows the monolayer WS₂ appearing as red lines cross-linking with the multilayer WS₂ (the dark region). Thus, the interconnected lines of the monolayer WS₂ contributed to the continuity of the as-prepared WS₂ films from the CVD method, which was consistent with the interconnected monolayers, as shown in the SEM images of Figure 3.

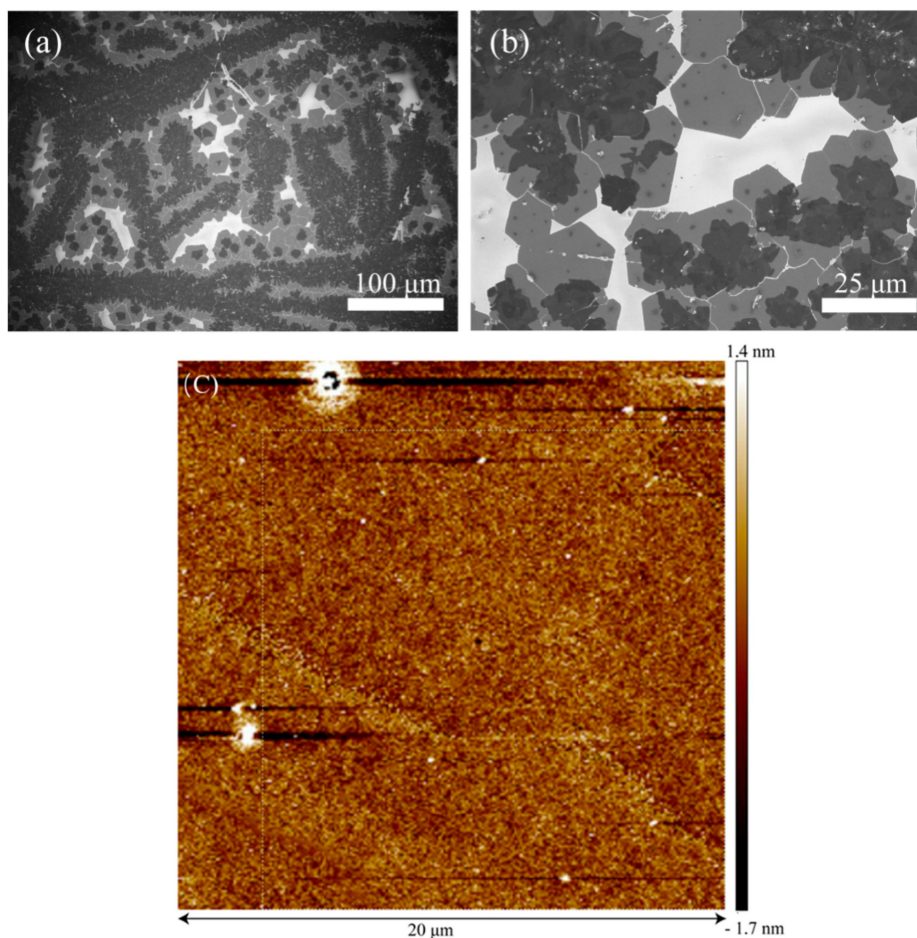


Figure 3. (a,b) Typical field emission scanning electron microscopy (FESEM) images of the WS₂ film. Images (a,b) were taken at different locations. (c) Atomic force microscope (AFM) image of 1:1 Poly-TPD and PCBM mixture.

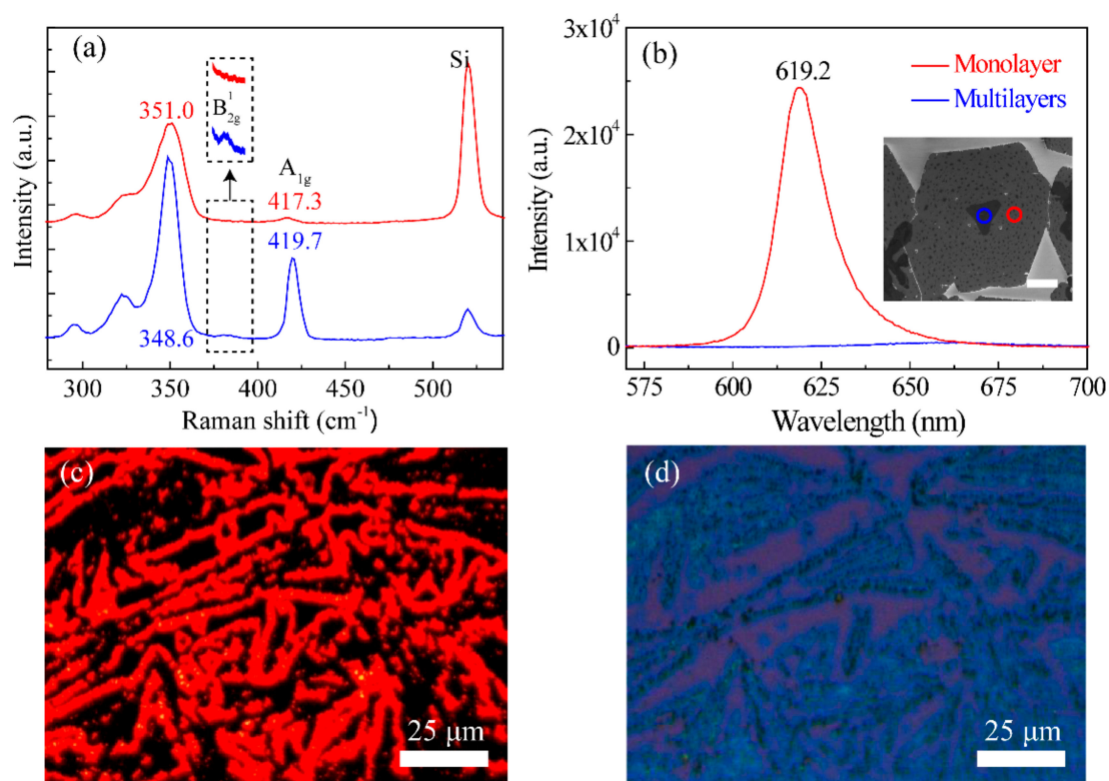


Figure 4. The Raman (a) and photoluminescence (PL) spectra (b) of the WS₂ monolayer (red line) and multilayers (blue line). Inset in (a) is a detailed illustration of the B_{2g}¹ and A_{1g} modes. The inset in (b) is an SEM image, and the scale bar represents 5 μm. (c) Photoluminescence (PL) and (d) optical microscopy images of WS₂ mono-/multimixed layers taken at the same location.

Opto-Electronic Properties of Photodetector

The linear scale of the *I*-*V* characteristics of the bilayer photodetectors was measured in the dark and under light with an intensity of 0.14 mW/cm² and a laser excitation wavelength of 450 nm (Figure 5a). The channel width of the device was 50 μm, and the length was 200 μm. The *I*-*V* curves showed a linear and symmetrical feature when no gate voltage was applied (*V*_G = 0 V), which confirmed that the as-fabricated device had ohmic contacts between the blend film and the Cr-Au electrodes. Figure 5b shows that the photocurrent increased quickly, with the gate voltage (*V*_G) sweeping from −30 V to 20 V. The responsivity (*R*) and detectivity (*D*^{*}) of the aforementioned photodetectors under illumination were calculated according to

$$R = \frac{I_{\text{light}} - I_{\text{dark}}}{PA}, \quad (1)$$

$$D^* = \frac{R}{\left(2e \frac{I_{\text{dark}}}{A}\right)^{1/2}}, \quad (2)$$

where *I*_{light} is the photocurrent, *I*_{dark} is the dark current, *P* is the light power density, *A* is the effective area of the photodetector, and *e* is the electronic charge. Figure 5c plots responsivity and detectivity, which increased with applied negative *V*_G. On the one hand, these indices rose as the applied voltage increased, and they grew linearly when the voltage was lower than −10 V, which suggests that not only was the photocurrent far larger than the dark current (Figure 5c), but also that a quite low power (voltage) input was required to amplify the photocurrent to the highest order of magnitude. With a −30 V bias voltage, the photocurrent (*I*_{light}) and dark current (*I*_{dark}) were 2.16 × 10^{−8} A and 7.56 × 10^{−9} A, respectively. Therefore, the as-fabricated 2D WS₂/Poly-TPD/PCBM/Au photodetector

achieved an R of 1.01 A/W and a high D^* of 1.4×10^{11} Jones at the 450-nm laser excitation ($V_G = -30$ V). The external quantum efficiency (EQE) was obtained by

$$EQE = \frac{Rhc}{e\lambda} \quad (3)$$

where hc/λ is the photon energy. The EQE (Figure 5d) exhibited the same gate voltage dependence trend as R , reaching about 3% at $V_G = -30$ V. As can be seen in Table 1, the obtained values in this work were comparable to those obtained in other work. Compared to WS_2 devices without organic materials [60,61], our device exhibited better performance, which indicated that the organic materials could improve the 2D material photodetector performance.

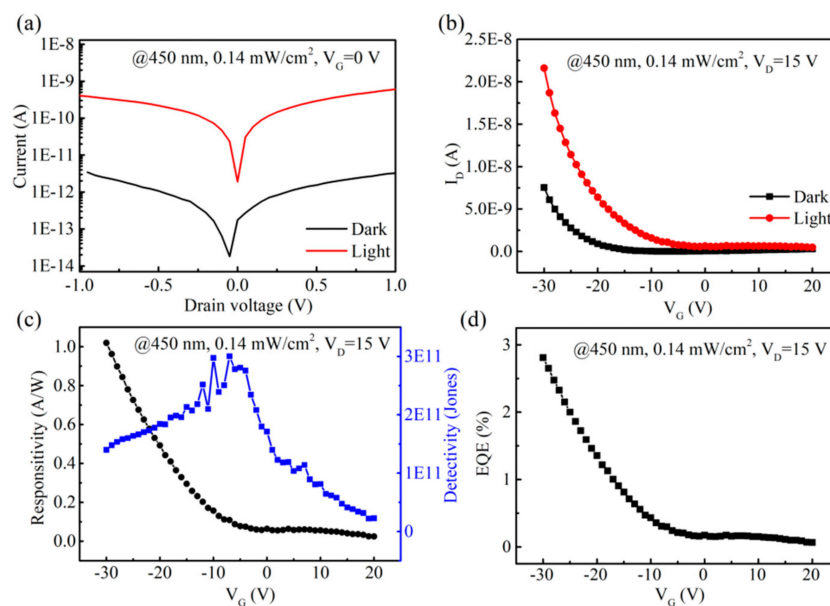


Figure 5. Performance of the WS_2 /Poly-TPD/PCBM-based photodetector. (a) Source-drain current (I_D) versus drain voltage (V_D). (b) I_D versus gate voltage (V_G) curves of the photodetector in dark (black line) and under laser (red line). (c,d) Responsivity and detectivity curves and the external quantum efficiency (EQE) of the device as a function of V_G ranging from -30 V to 20 V. Measurements were performed at room temperature with a laser at 450 nm. The light intensity was 0.14 mW/cm 2 . The V_G was 0 V in (a), and the V_D was 15 V in (b–d).

Table 1. Photoresponse parameters of different materials and devices.

Device Structure	Incident Light	R (A/W)	D (Jones)	Response Time	Ref.
2D WS_2 /Poly-TPD/PCBM/Au	450 nm	1.02	1.4×10^{11}	<181 ms	This work
Monolayer WS_2 /Au	532 nm	0.59	6.5×10^{10}	280 ms	[60]
Multilayer WS_2 /Au	635 nm	0.7	2.7×10^{10}	4.1 s	[61]
MoTe $_2$ /Au	532 nm	0.0004	1.08×10^8	42.5 μ s	[62]
GeSe $_2$ /Au	450 nm	2.5	N/A	0.2 s	[63]
GeP/Au	532 nm	3.1	N/A	>1s	[64]

Figure 6a shows the photocurrent versus applied V_D sweeping from -10 V to 10 V. In addition, 10 different illumination intensities were applied to the WS_2 /organic device with varied laser intensities ranging from 0.1 mW/cm 2 to 0.55 mW/cm 2 . These I – V plots showed a linear increase of the photocurrent with the applied voltage. In Figure 6b, the dependence of the photocurrent ($I_{D,light}$) (in Figure 6a) on the laser intensity (mW/cm 2) was plotted on a log scale ($V_D = 10$ V). The $I_{D,light}$ increased gradually with power intensity when the power was 0.4 mW/cm 2 and then showed a saturating tendency with higher power intensity. As shown in Figure 6c, the responsivity R increased nearly linearly to 17 A/W

with incident laser power when the power intensity reached 0.4 mW/cm^2 , and then it was gradually saturated with further increased power intensity ($V_D = 10 \text{ V}$).

The broad-spectrum response of the as-prepared bilayer photodetector was investigated with an incident laser with a wavelength varying from 500 nm to 750 nm (Figure 6d). Herein, the EQE was calculated from the I - V curves ($V_D = 10 \text{ V}$ and $V_G = 0 \text{ V}$), and the incident light was a 100-ps pulse laser with a 4-MHz frequency. The peaks in Figure 6d corresponded well to the A exciton ($\lambda = 626 \text{ nm}$), B exciton ($\lambda = 527 \text{ nm}$), and indirect band gap (I) absorption, as shown in Figure 2c, indicating that the effective absorption in the WS_2 thin films actively contributed to the photocurrent generation [24,51,52]. As shown in Figure 1d, the LUMO at -4.2 eV of PCBM and the HOMO at -5.2 eV of Poly-TPD were well aligned with the conduction and valence levels of WS_2 to dissociate the excitons typically generated inside the WS_2 films, with an exciton binding energy of 0.3 eV to 0.7 eV [15,16]. The reliable and rapid response speed of the bilayer photodetector was examined with pulsed laser illumination, as shown in Figure 6e,f. The PD promptly responded to 450 nm of light at the millisecond level ($<181 \text{ ms}$), giving rise to a sharply enhanced and decayed photocurrent upon multiple switching cycles.

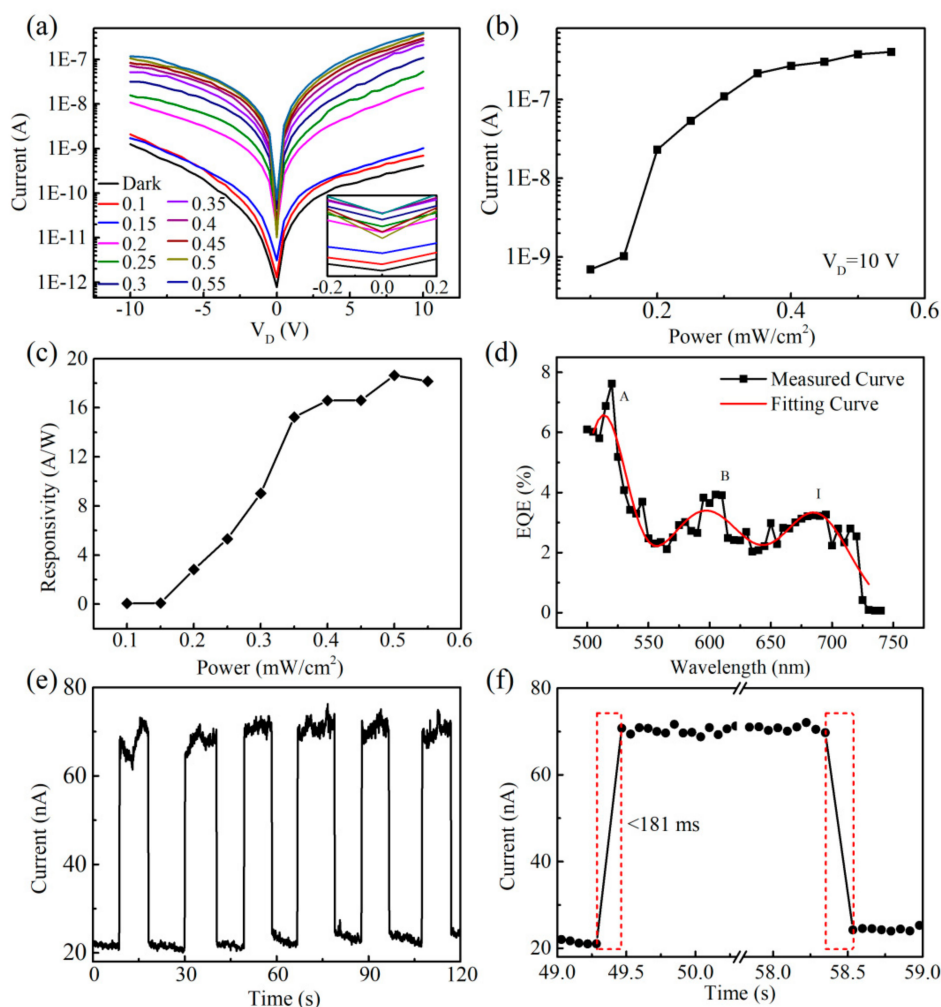


Figure 6. (a) The photocurrent versus applied V_D sweeping from -10 V to 10 V ($V_G = 0 \text{ V}$) at different light intensities. The inset shows the detailed I - V characteristics in the 0 V region. (b) The dependence of the photocurrent ($I_{D/light}$) (in (a)) on the laser intensity (mW/cm^2) ($V_G = 0 \text{ V}$, $V_D = 10 \text{ V}$). (c) Power intensity dependence of the responsivity ($V_G = 0 \text{ V}$, $V_D = 10 \text{ V}$). (d) Spectral dependence of the EQE using a 100-ps pulse laser ($V_G = 0 \text{ V}$, $V_D = 10 \text{ V}$). (e) Photoresponse in the dark and under 450-nm laser irradiation with a light intensity of 0.14 mW/cm^2 ($V_G = 0 \text{ V}$, $V_D = 10 \text{ V}$). (f) Rise and decay time of the photodetector ($V_G = 0 \text{ V}$, $V_D = 10 \text{ V}$).

The 2D WS₂/Poly-TPD/PCBM bilayer photodetectors with few layers of WS₂ could detect different photon wavelengths and a wide range of incident intensities, making it a strong candidate for constructing novel optoelectronic devices. We attribute the excellent photoresponse to the appropriate band gap, the high quality of the CVD-grown single/multilayer WS₂, and the energy-favorable heterojunction. The as-prepared WS₂ had excellent photon absorption throughout a wide range of 500 nm to 750 nm, allowing enough exciton (or electron-hole pair) generation. As shown in Figure 1d, 2D WS₂ (conductive band edge at ca. −3.84 eV and valence band edge at ca. −5.82 eV for the monolayer) was a perfect match with the Poly-TPD/PCBM mixture (LUMO at −4.1 eV of PCBM to accept electrons and HOMO at −5.2 eV of Poly-TPD to accept holes). The energy offset of the WS₂ and organic blend film (the offset between WS₂ and Poly-TPD was 0.62 eV and between WS₂ and PCBM was 0.26 eV) was sufficiently high enough to dissociate excitons generated from monolayer WS₂, since the typical exciton binding energy was around 0.71 ± 0.01 eV [15]. Therefore, Poly-TPD was favorable for hole transport and PCBM for electron transport to their respective electrodes. Notably, Poly-TPD and PCBM were intermixed throughout the entire organic blend film, and it is highly possible that portions of the electron and hole carriers recombined first before arriving at their electrodes, resulting in a relatively low EQE when the $V_G = 0$ V (Figure 5d). Therefore, the built-in electric field needed to be applied through the gate voltage, and more photogenerated electrons and holes could quickly drift away to their electrodes. As shown in Figure 5d, with the increase in the gate voltage applied, the current increased exponentially, which is typical diode behavior. As a result, the EQE could be modulated to 3% at $V_G = -30$ V.

Herein, the limited trap state in the CVD-grown WS₂ layer also greatly reduced the exciton quenching, and photoconductivity might have dominated the fast response time, which was consistent with the nearly linear dependence of the photocurrent on incident laser power intensity (Figure 6b). However, photocurrents higher than 0.1 μ A were gradually saturated with further increased laser power intensity, indicating that the charge collection efficiency caused a limiting factor due to relatively low carrier mobility in organic blend films. Correspondingly, chances for electron and hole-free carriers under higher irradiation excitations to recombine increased before collection by their respective Au electrodes. Notably, this saturated photocurrent behavior was quite different from pure 2D WS₂-sandwiched PDs, which typically have nonlinear dependence on the incident laser power intensity $P^{0.5}$ [65]. As a result, the responsivity of the as-prepared bilayer 2D-WS₂/organic photodetector had a saturated trend quite similar with the radiant power intensity, reaching a maximum of 17 A/W. The wide visible spectrum response (Figure 6d) clearly showed that single/multilayers of WS₂ laminated with organic blend films is a promising device architecture for high-performance optoelectronic applications. Optimization of the organic blend layer conductivity will further improve this unique 2D/organic bilayer photodetector performance.

4. Conclusions

In summary, we prepared an inorganic/organic heterojunction photodetector through highly oriented 2D WS₂ film and Poly-TPD/PCBM blend organic films. The results revealed that the photodetector had high responsivity and detectivity at room temperature, exhibiting a high detectivity of 1.1×10^{11} Jones at zero-gate voltage and a responsivity of 17 A/W. The spin-coated organic films effectively smoothed out the WS₂ films, forming an energy barrier to significantly suppress the dark current. The on/off ratio of these bilayer PDs was as high as 100, and a rise time of less than 181 ms was obtained, indicating fast electron/hole dissociation at the interface of the 2D WS₂ and Poly-TPD/PCBM (1:1) organic layers. We attribute the excellent photoresponse to the appropriate band gap, the high quality of the CVD-grown single/multilayer WS₂, and the energy-favorable heterojunction for efficient exciton separation. Further improvement on carrier mobility in organic layers will enhance the charge collection efficiency. Our work offers a solution-coating process combined with a CVD technique that can create high-crystalline 2D films for high-performance photodetectors over regular silicon

substrates. The inorganic 2D/organic heterostructures formed will create a new and fruitful paradigm in optoelectronics.

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